

Multi-scale Modelling of Complex Materials undergoing Strain Gradients and Damage

Severo Ochoa Seminar

Igor A. Rodrigues Lopes
ilopes@inegi.up.pt

INTRODUCTION

Research experience

Senior researcher at INEGI



2013 – MSc in Mechanical Engineering (FEUP)

- Multi-scale analysis of porous materials

2014-2019 – PhD in Mechanical Engineering (FEUP)

- Second-order computational homogenization (strain-gradients)
- Implementation of FE2 analyses in in-house code
- Parallel computing
- Lagrange multiplier method for more efficient multi-scale analysis

2020-2022 – TREAL - CleanSky2 project (FEUP)

- Finite strain visco-elastic – visco-plastic model for composites
- Finite strain smeared crack model for composites

2022-2026 – DIDEAROT - HORIZON project (INEGI)

- Damage models for composites (micro and meso-scales)
- Surrogate models for composites



Other Interests

- Multi-scale modelling of fracture
- Second-order homogenization for metamaterials and damage

ABOUT INEGI / FEUP / U.Porto



Founded in 1986



9 267 m²

Facilities - Porto



INNOVATION AND
TECHNOLOGY
TRANSFER

RESEARCH &
DEVELOPMENT

SERVICES

Consulting
Laboratory Services
Advanced Training

NEW COMPOSITE MATERIALS AND PROCESSES

METALLOGRAPHY AND ADVANCED CASTING PROCESSES

METAL FORMING

ADDITIVE MANUFACTURING

COMPUTATIONAL MECHANICS

STRUCTURAL HEALTH MONITORING

TRIBOLOGY, VIBRATIONS AND DYNAMICS

MECHANICAL DESIGN

INSTRUMENTATION, AUTOMATION AND CONTROL

PRODUCT DEVELOPMENT

BIOMECHANICS

WIND ENERGY

ENERGY EFFICIENCY

OPERATIONS AND SUPPLY CHAIN MANAGEMENT

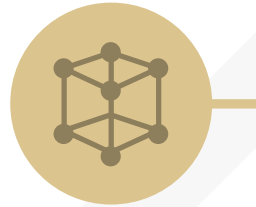
Constitutive modelling of deformation

Ductile metallic materials

Amorphous polymeric materials

Hexagonal compact packed materials

Fibre reinforced polymeric materials



Links

*Large Strain Implicit Non-Linear
Analysis of Solids Linking Scales*



Multiscale modelling

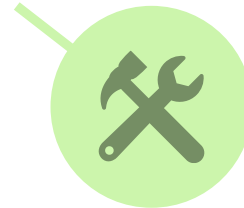
Rough contact interfaces

Micromechanical analysis

Polycrystalline alloys

Second-order effects

Coupled analysis



Computational methods

Contact modelling

Incompressible deformation

Non-local ductile damage

High performance computing



CM2S

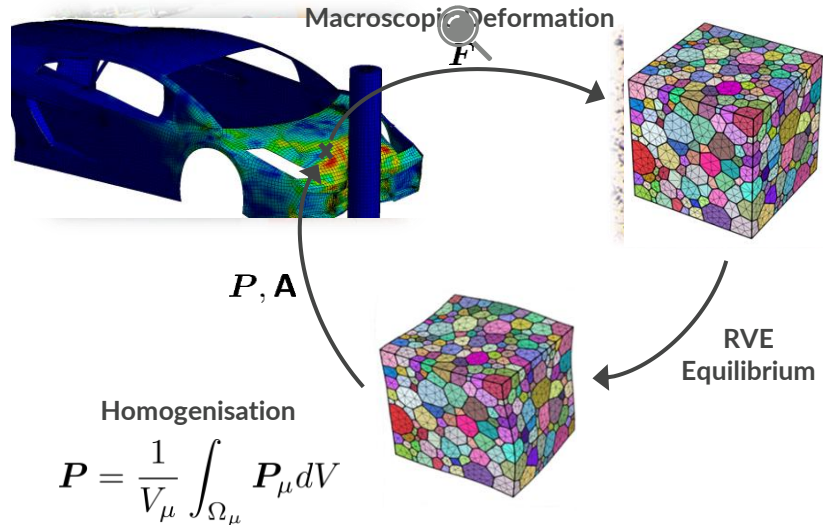
Computational Multi-Scale
Modeling of Solids and Structures

U. PORTO
FEUP FACULDADE DE ENGENHARIA
UNIVERSIDADE DO PORTO

COMPUTATIONAL HOMOGENISATION



COMPUTATIONAL HOMOGENISATION CONCEPT



A computationally efficient solution of the RVE problem is critical in these applications, since all of them rely on a large number of micro-scale simulations!

A **Representative Volume Element (RVE)** is used to model the microstructure.

Only **constitutive models** for individual micro-constituents are needed. The resulting **compound response is naturally accounted for**.

Suitable for **arbitrary material behaviour** and geometrical **evolution of the microstructure**.

Enables the incorporation of **complex microstructural geometries** together with **large deformations**.

Applications:

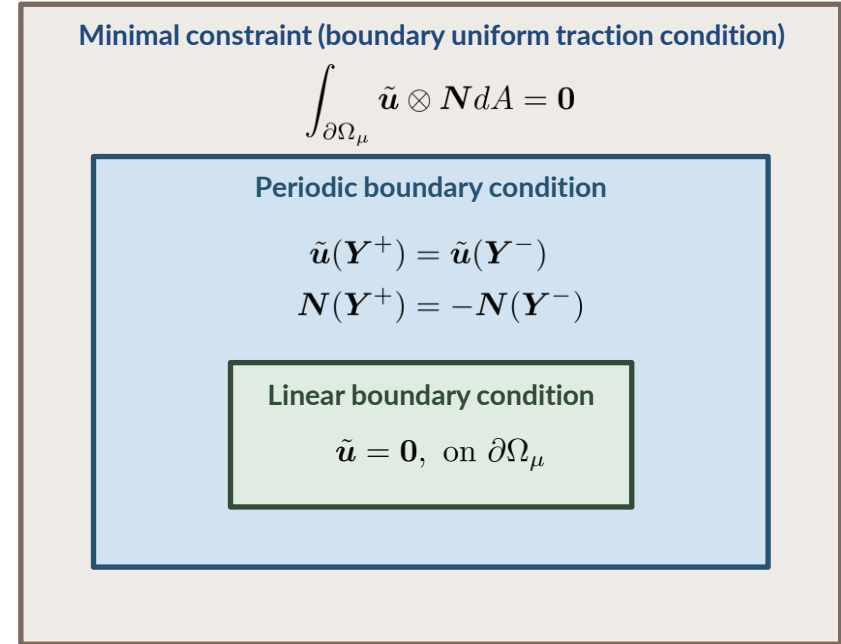
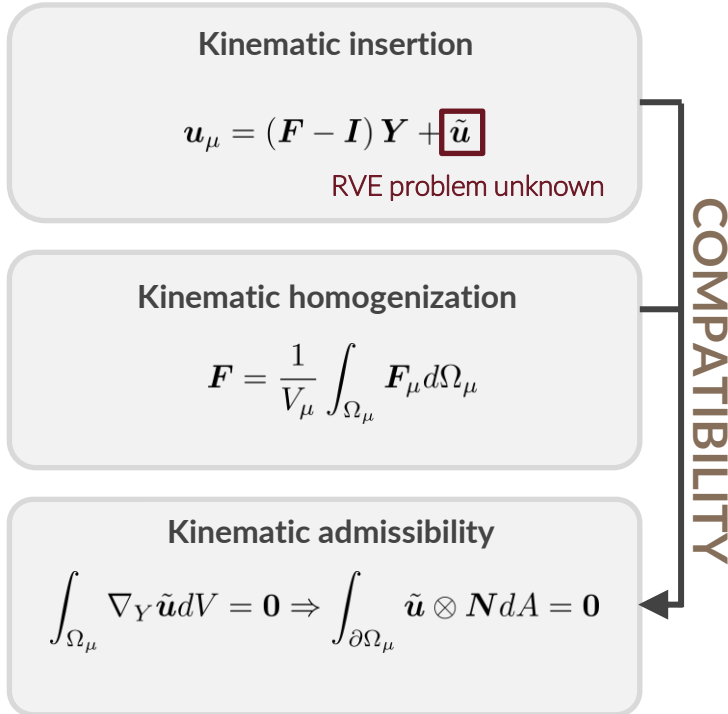
- **FE² simulations**
- **Parametric studies** (for instance, to generate macroscopic yield surfaces)
- **Topology optimization**
- **Response database** to build **reduced-order models** or to apply in **data-driven frameworks**

COMPUTATIONAL HOMOGENISATION

KINEMATIC CONSTRAINTS

Following the Method of Multi-Scale Virtual Power

[Blanco et al. (2016), *Arch Computat Methods Eng* 23(2) 191-253]



Spaces of admissible solutions:
 $\tilde{\mathcal{V}}^{linear} \subset \tilde{\mathcal{V}}^{periodic} \subset \tilde{\mathcal{V}}^{u.traction}$

MICRO-SCALE EQUILIBRIUM PROBLEM – CONDENSATION METHOD

Hill-Mandel Principle

$$\mathbf{P} : \delta \mathbf{F} = \frac{1}{V_\mu} \int_{\Omega_\mu} \mathbf{P}_\mu : (\delta \mathbf{F} + \nabla_Y \delta \tilde{\mathbf{u}}) dV$$

$$\forall \delta \mathbf{F}, \delta \tilde{\mathbf{u}} \in \delta \tilde{\mathcal{V}} \text{ and } \tilde{\mathbf{u}} \in \tilde{\mathcal{V}}$$



$$\delta \mathbf{F} = \mathbf{0}$$

Micro-Scale Weak Equilibrium

$$\int_{\Omega_\mu} \mathbf{P}_\mu : \nabla_Y \delta \tilde{\mathbf{u}} dV = 0, \quad \forall \tilde{\mathbf{u}} \in \tilde{\mathcal{V}}$$



Finite Element Solution

$$\mathbf{K} \Delta \tilde{\mathbf{u}} = -\mathbf{r}$$

Linear boundary condition

$$\left[\mathbf{k}^{ii} \right] \left\{ \Delta \tilde{\mathbf{u}}^i \right\} = - \left\{ \mathbf{f}^i \right\}$$

(boundary dofs elimination)

Periodic boundary condition
(conform mesh)

$$\left[\begin{array}{c|c} \mathbf{k}^{ii} & \mathbf{k}^{i+} + \mathbf{k}^{i-} \\ \hline \mathbf{k}^{+i} + \mathbf{k}^{-i} & \mathbf{k}^{++} + \mathbf{k}^{-+} + \mathbf{k}^{+-} + \mathbf{k}^{--} \end{array} \right] \left\{ \begin{array}{c} \Delta \tilde{\mathbf{u}}^i \\ \Delta \tilde{\mathbf{u}}^+ \end{array} \right\} = - \left\{ \begin{array}{c} \mathbf{f}^i \\ \mathbf{f}^+ + \mathbf{f}^- \end{array} \right\}$$

(boundary dofs standard condensation)

Uniform traction condition
(results in multi-point constraints)

$$\int_{\partial \Omega_\mu} \tilde{\mathbf{u}} \otimes \mathbf{N} dA = \mathbf{0} \Rightarrow \mathbf{C} \tilde{\mathbf{u}} = \mathbf{0} \quad \Delta \tilde{\mathbf{u}} = \left\{ \begin{array}{c} \Delta \tilde{\mathbf{u}}^i \\ \Delta \tilde{\mathbf{u}}^f \\ \Delta \tilde{\mathbf{u}}^d \\ \Delta \tilde{\mathbf{u}}^p \end{array} \right\} \quad \Delta \tilde{\mathbf{u}}^p = \mathbf{0} \quad \Delta \tilde{\mathbf{u}}^d = \alpha \Delta \tilde{\mathbf{u}}^f$$

$$\alpha = [\mathbf{C}^d]^{-1} \mathbf{C}^f$$

$$\left[\begin{array}{c|c} \mathbf{k}^{ii} & \mathbf{k}^{if} + \mathbf{k}^{id} \alpha \\ \hline \mathbf{k}^{fi} + \alpha^T \mathbf{k}^{di} & \mathbf{k}^{ff} + \alpha^T \mathbf{k}^{df} + \mathbf{k}^{fd} \alpha + \alpha \mathbf{k}^{dd} \alpha^T \end{array} \right] \left\{ \begin{array}{c} \Delta \tilde{\mathbf{u}}^i \\ \Delta \tilde{\mathbf{u}}^f \end{array} \right\} = - \left\{ \begin{array}{c} \mathbf{f}^i \\ \mathbf{f}^f + \alpha^T \mathbf{f}^d \end{array} \right\}$$

COMPUTATIONAL HOMOGENISATION

MICRO-SCALE EQUILIBRIUM PROBLEM – LAGRANGE MULTIPLIER METHOD

Hill-Mandel Principle

$$\begin{aligned} \mathbf{P} : \delta \mathbf{F} &= \frac{1}{V_\mu} \left[\int_{\Omega_\mu} \mathbf{P}_\mu : (\delta \mathbf{F} + \nabla_Y \delta \tilde{\mathbf{u}}) dV \right. \\ &\quad \left. - \delta \mathbf{L} : \left(\int_{\partial \Omega_\mu} \tilde{\mathbf{u}} \otimes \mathbf{N} dA \right) - \mathbf{L} : \left(\int_{\partial \Omega_\mu} \delta \tilde{\mathbf{u}} \otimes \mathbf{N} dA \right) \right] \\ &\quad \forall (\delta \mathbf{F}, \delta \tilde{\mathbf{u}}, \delta \mathbf{L}) \end{aligned}$$

$$\delta \mathbf{F} = \mathbf{0}$$

Micro-Scale Weak Equilibrium

$$\begin{aligned} \int_{\Omega_\mu} \mathbf{P}_\mu : \nabla_Y \delta \tilde{\mathbf{u}} dV - \delta \mathbf{L} : \left(\int_{\partial \Omega_\mu} \tilde{\mathbf{u}} \otimes \mathbf{N} dA \right) \\ - \mathbf{L} : \left(\int_{\partial \Omega_\mu} \delta \tilde{\mathbf{u}} \otimes \mathbf{N} dA \right) = 0 \end{aligned}$$

Finite Element Solution

$$\begin{bmatrix} \mathbf{k}^{ii} & \mathbf{k}^{if} & \mathbf{0} \\ \mathbf{k}^{fi} & \mathbf{k}^{ff} & \mathbf{C}_L^T \\ \mathbf{0} & \mathbf{C}_L & \mathbf{0} \end{bmatrix} \begin{Bmatrix} \Delta \tilde{\mathbf{u}}^i \\ \Delta \tilde{\mathbf{u}}^f \\ \Delta \lambda_L \end{Bmatrix} = - \begin{Bmatrix} \mathbf{f}^i \\ \mathbf{f}^f - \mathbf{C}_L^T \lambda_L \\ \mathbf{C}_L \tilde{\mathbf{u}}^f \end{Bmatrix}$$

Homogenised Stress

$$\mathbf{P} = \frac{1}{V_\mu} \int_{\Omega_\mu} \mathbf{P}_\mu dV = \mathbf{L}$$

Macroscopic consistent tangent

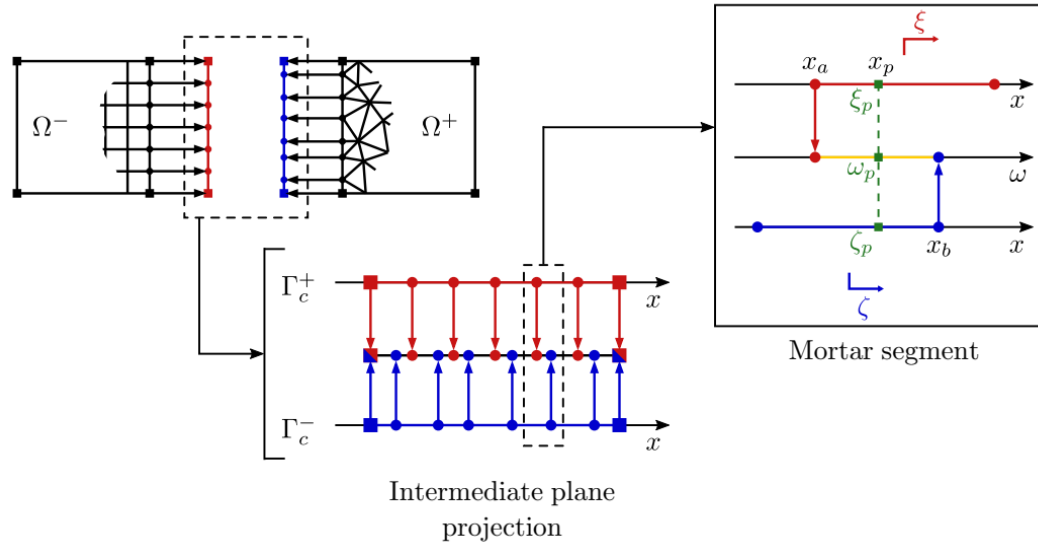
$$\mathbf{A} = \frac{\partial \mathbf{P}}{\partial \mathbf{F}} = \frac{\partial \lambda_L}{\partial \mathbf{F}}$$

$$\begin{bmatrix} \mathbf{k}^{ii} & \mathbf{k}^{if} & \mathbf{0} \\ \mathbf{k}^{fi} & \mathbf{k}^{ff} & \mathbf{C}_L^T \\ \mathbf{0} & \mathbf{C}_L & \mathbf{0} \end{bmatrix} \begin{bmatrix} \frac{\partial \tilde{\mathbf{u}}^i}{\partial \mathbf{F}} \\ \frac{\partial \tilde{\mathbf{u}}^f}{\partial \mathbf{F}} \\ \frac{\partial \lambda_L}{\partial \mathbf{F}} \end{bmatrix} = - \begin{bmatrix} \mathbf{k}^{ii} & \mathbf{k}^{if} & \mathbf{k}^{ip} \\ \mathbf{k}^{fi} & \mathbf{k}^{ff} & \mathbf{k}^{fp} \\ \mathbf{0} & \mathbf{0} & \mathbf{0} \end{bmatrix} \begin{bmatrix} \mathbf{D}^i & \mathbf{D}^f & \mathbf{D}^p \end{bmatrix}$$

COMPUTATIONAL HOMOGENISATION

MORTAR PERIODIC BOUNDARY CONDITION

I. A. Rodrigues Lopes, B. P. Ferreira, and F. M. Andrade Pires (2021), *Comput. Methods Appl. Mech. Eng.*, vol. 384, p. 113930



Micro-Scale Weak Equilibrium

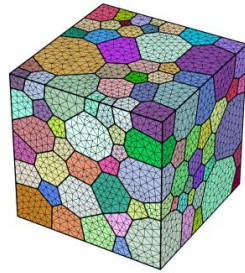
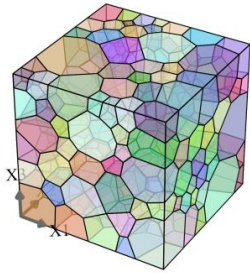
$$\int_{\Omega_\mu} \mathbf{P}_\mu : \nabla_Y \delta \tilde{\mathbf{u}} dV + \int_{\partial\Omega_\mu^+} \delta \boldsymbol{\lambda} \cdot (\tilde{\mathbf{u}}^+ - \tilde{\mathbf{u}}^-) dA + \int_{\partial\Omega_\mu^+} \boldsymbol{\lambda} \cdot (\delta \tilde{\mathbf{u}}^+ - \delta \tilde{\mathbf{u}}^-) dA = 0$$

Finite Element Solution

$$\begin{bmatrix} \mathbf{k}^{ii} & \mathbf{k}^{i-} & \mathbf{k}^{i+} & \mathbf{0} \\ \mathbf{k}^{-i} & \mathbf{k}^{--} & \mathbf{k}^{-+} & -[\mathbf{A}^m]^T \\ \mathbf{k}^{+i} & \mathbf{k}^{+-} & \mathbf{k}^{++} & \mathbf{D}^{nm} \\ \mathbf{0} & -\mathbf{A}^m & \mathbf{D}^{nm} & \mathbf{0} \end{bmatrix} \begin{Bmatrix} \Delta \tilde{\mathbf{u}}^i \\ \Delta \tilde{\mathbf{u}}^- \\ \Delta \tilde{\mathbf{u}}^+ \\ \Delta \boldsymbol{\lambda} \end{Bmatrix} = - \begin{Bmatrix} \mathbf{f}^i \\ \mathbf{f}^- - [\mathbf{A}^m]^T \boldsymbol{\lambda} \\ \mathbf{f}^+ + \mathbf{D}^{nm} \boldsymbol{\lambda} \\ \mathbf{D}^{nm} \tilde{\mathbf{u}}^+ - \mathbf{A}^m \tilde{\mathbf{u}}^- \end{Bmatrix}$$

COMPUTATIONAL HOMOGENISATION

EFFICIENCY OF THE LAGRANGE MULTIPLIER APPROACH



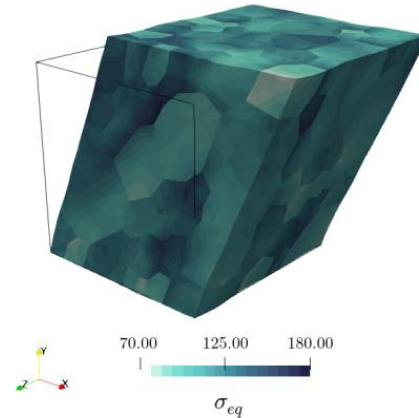
250 grains
62248 tetra 4 elements
11908 nodes

Anisotropic Saint Venant-Kirchhoff hyperelasticity:

$D_{11} = 168.4$ GPa, $D_{12} = 121.4$ GPa, and $D_{44} = 75.4$ GPa

$$\mathbf{D}^e = \begin{bmatrix} D_{11} & D_{12} & D_{12} & 0 & 0 & 0 \\ & D_{11} & D_{12} & 0 & 0 & 0 \\ & & D_{11} & 0 & 0 & 0 \\ & & & D_{44} & 0 & 0 \\ \text{sym.} & & & & D_{44} & 0 \\ & & & & & D_{44} \end{bmatrix}$$

$$\mathbf{F} = \begin{bmatrix} 1.0 & 0.5 & 0.0 \\ 0.0 & 1.2 & 0.0 \\ 0.0 & 0.0 & 1.0 \end{bmatrix}$$

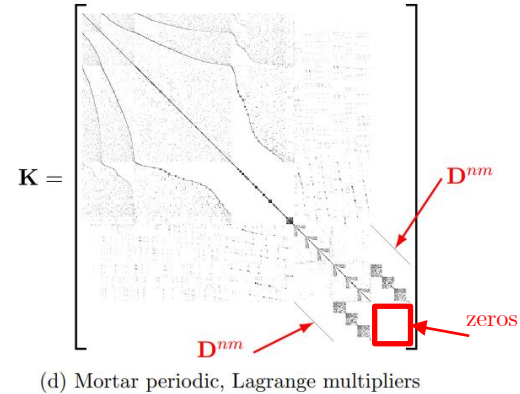
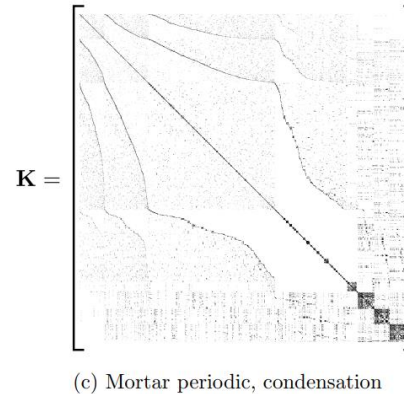
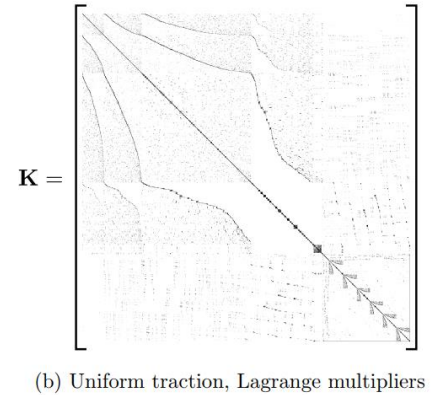
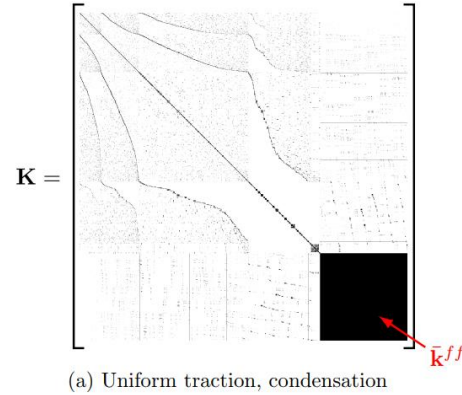
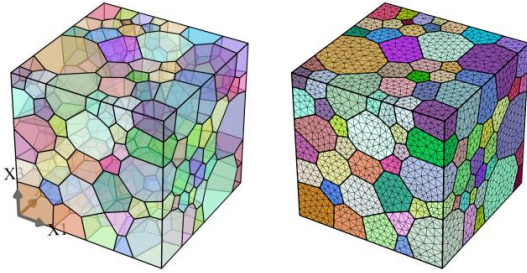


		Time (s)		Speedup
		Cond.	L.M.	
Uniform traction	solver	140.44	3.45	40.71
	iteration	146.90	3.96	37.10
	total	7647	232	32.90
Mortar periodic	solver	10.67	13.03	0.82
	iteration	99.79	13.29	7.51
	total	4812	669	7.19

	Mem. peak (MB)		Savings (%)
	Cond.	L.M.	
Uniform traction	9492	761	92.0
Mortar periodic	6046	1238	79.5

COMPUTATIONAL HOMOGENISATION

EFFICIENCY OF THE LAGRANGE MULTIPLIER APPROACH



	Sparsity (%)	
	Cond.	L.M.
Uniform traction	7.8075	0.7037
Mortar periodic	0.9994	0.5526

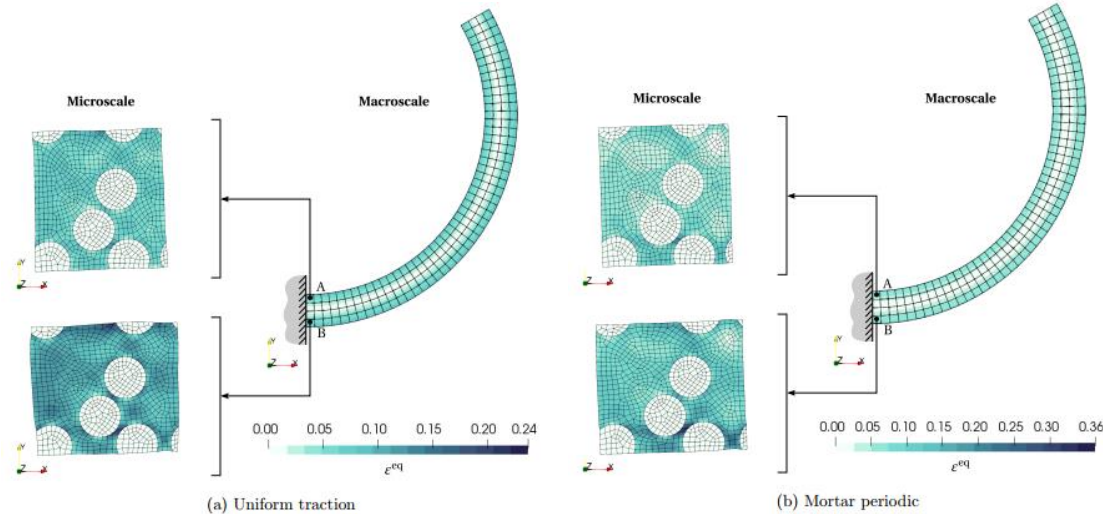
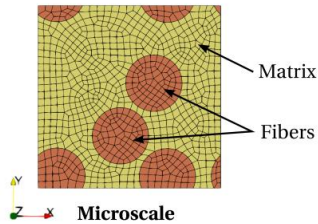
COMPUTATIONAL HOMOGENISATION

EFFICIENCY OF THE LAGRANGE MULTIPLIER APPROACH

200 elements with 4 Gauss points: 800 RVE simulations in each macro-iteration
 Parallelization with 16 CPUs [Rodrigues Lopes et al. (2018) Comp.Mech. 61 (1-2) 157-180]

	Matrix	Inclusions
Young modulus (E) [GPa]	2.0	74.0
Poisson ratio (ν)	0.4	0.2

RVE
 896 quad8 elements
 2793 nodes



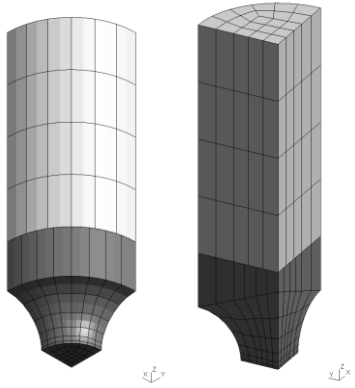
		Time (s)		Speedup
		Cond.	L.M.	
Uniform traction	solver	0.097	0.067	1.45
	iteration	0.132	0.089	1.48
	total	5.54	3.74	1.48
Mortar periodic	solver	0.063	0.083	0.76
	iteration	0.126	0.104	1.21
	total	5.16	4.20	1.22

		Time (s)		Speedup
		Cond.	L.M.	
Uniform Traction	iteration	37.3	15.3	2.44
	total	5039	2007	2.51
	Periodic Mortar	iteration	31.5	17.1
	total	4371	2280	1.92

COMPUTATIONAL HOMOGENISATION

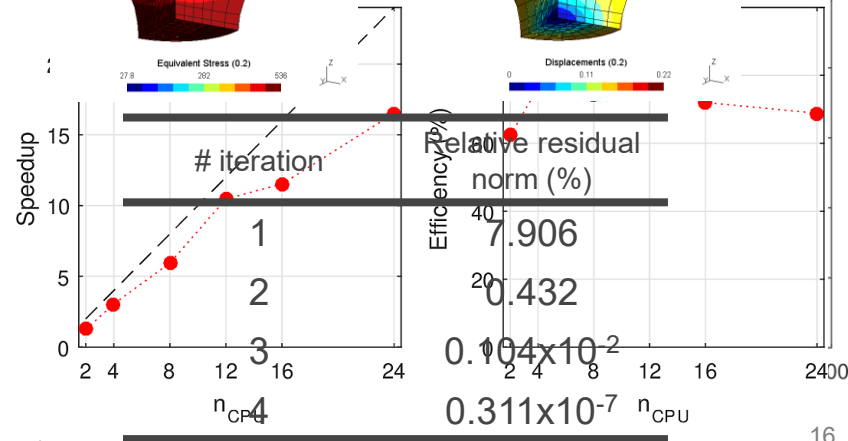
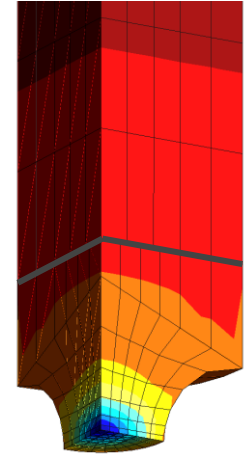
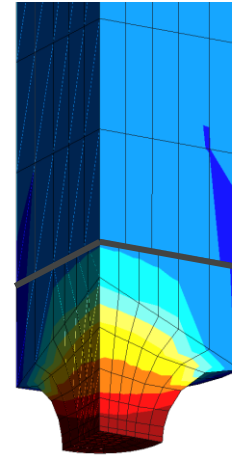
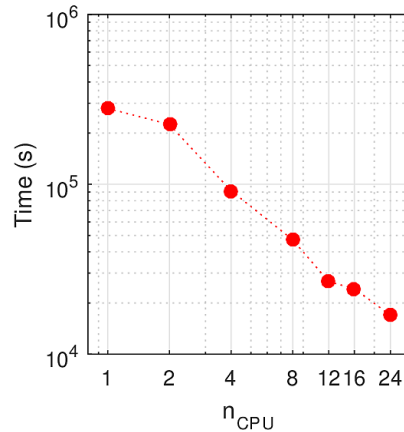
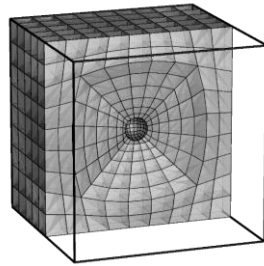
PARALLEL COMPUTING

MACRO



84 Hexa20 light grey elements
224 Hexa20 dark grey elements

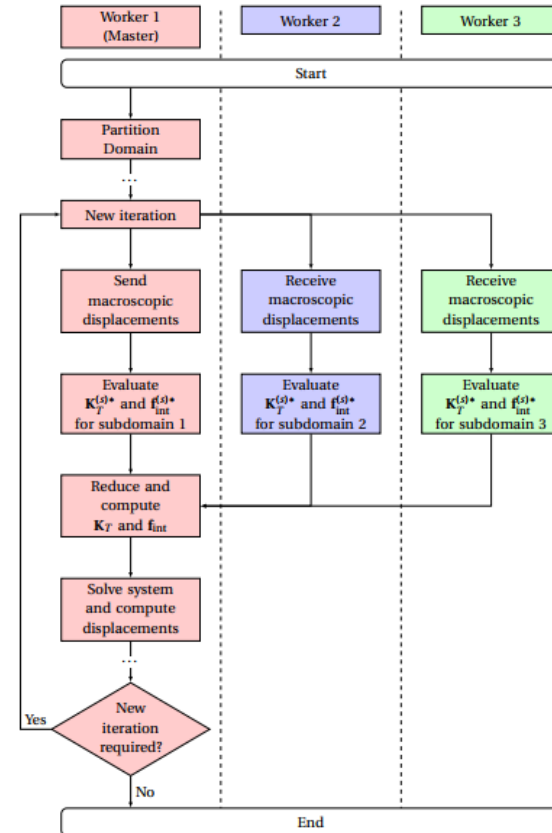
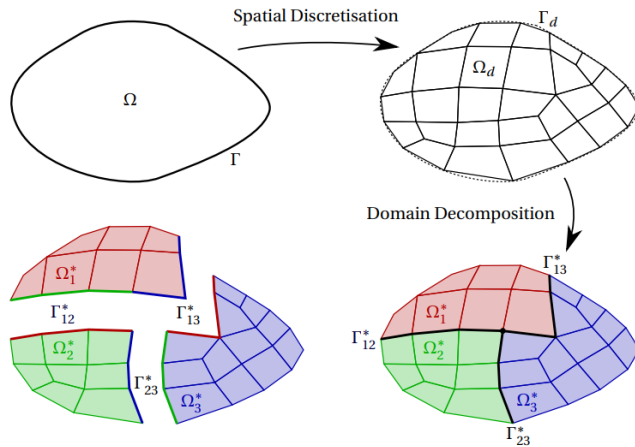
MICRO



COMPUTATIONAL HOMOGENISATION

PARALLEL COMPUTING

[Rui Coelho (2021), Internal Report]

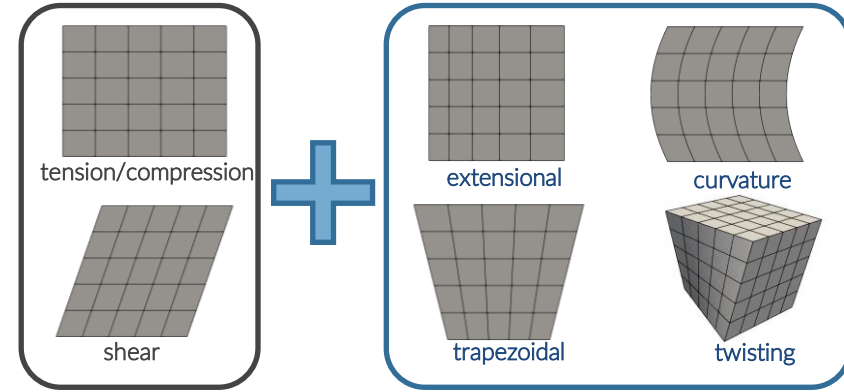
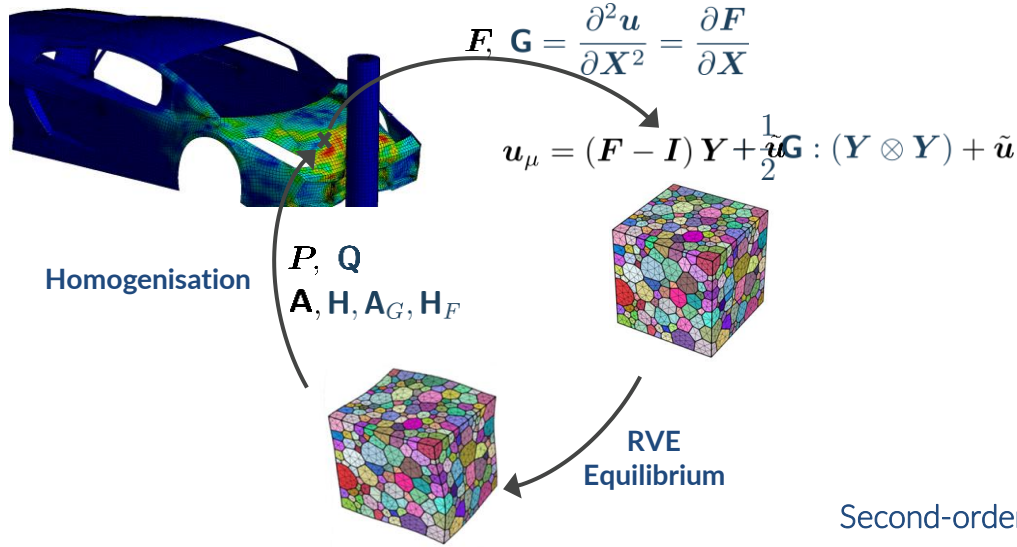


COMPUTATIONAL HOMOGENISATION WITH STRAIN GRADIENTS



SECOND-ORDER COMPUTATIONAL HOMOGENISATION

CONCEPT



Second-order homogenisation:

- Second-gradient continuum at the macro-scale
- Linear variation of the macro-deformation gradient
- Second-order deformation modes at the RVE
- The RVE length results in a length scale parameter
- Captures size effects

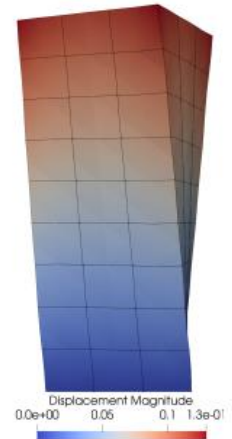
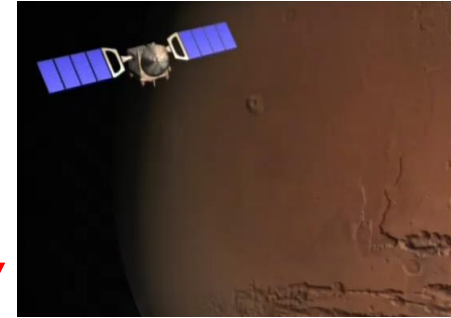
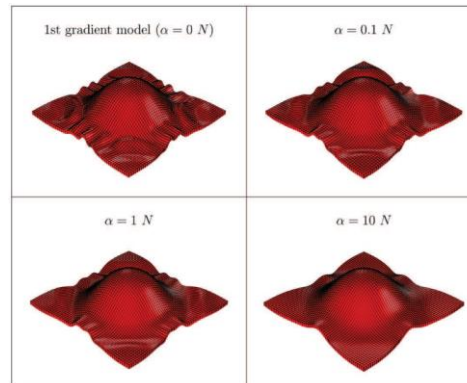
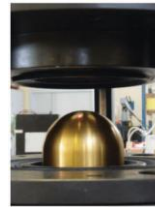
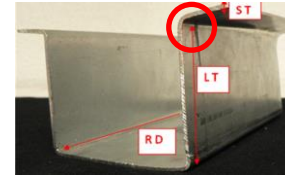
SECOND-ORDER COMPUTATIONAL HOMOGENISATION

POTENTIAL APPLICATIONS



Interesting applications for second-order homogenisation:

- Analysis and design of structures subjected to deformations with high curvatures
- Metal sheet forming
- Design of actuators based on phase transformation
- Deformation of woven composites
- Modelling the behaviour meta-materials



SECOND-ORDER COMPUTATIONAL HOMOGENISATION

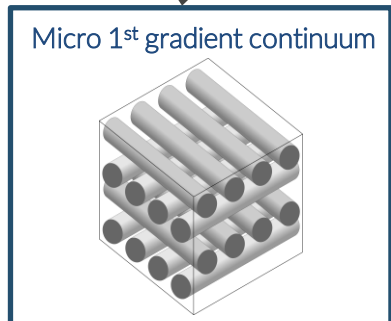
FORMULATION

Formulation detailed in: Rodrigues Lopes, I. A., & Andrade Pires, F. M. (2022). *Comput. Methods in Appl. Mech. Engrg*, 392, 114714.

Review in: Rodrigues Lopes, I. A., & Andrade Pires, F. M. (2022). *Arch. Comput. Methods Eng.*, 29(3), pp. 1339–1393



\mathbf{F}, \mathbf{G}



POSTULATED

Kinematic insertion

$$\mathbf{u}_\mu = (\mathbf{F} - \mathbf{I}) \mathbf{Y} + \frac{1}{2} \mathbf{G} : (\mathbf{Y} \otimes \mathbf{Y}) + \tilde{\mathbf{u}}$$

Kinematic homogenisation

$$\mathbf{F} = \frac{1}{V_\mu} \int_{\Omega_\mu} \mathbf{F}_\mu d\Omega_\mu$$

$$\mathbf{G} = \frac{1}{V_\mu} \int_{\Omega_\mu} (\nabla_Y \mathbf{u}_\mu \otimes \mathbf{Y}) \cdot \mathbf{J}^{-1} d\Omega_\mu$$

Kinematic admissibility

$$1^{\text{st}} \text{ Constraint: } \int_{\Omega_\mu} \nabla_Y \tilde{\mathbf{u}} dV = \mathbf{0} \Rightarrow \int_{\partial\Omega_\mu} \tilde{\mathbf{u}} \otimes \mathbf{N} dA = \mathbf{0}$$

$$2^{\text{nd}} \text{ Constraint: } \int_{\Omega_\mu} (\nabla_Y \tilde{\mathbf{u}} \otimes \mathbf{Y}) \cdot \mathbf{J}^{-1} dV = \mathbf{0}$$

COMPATIBILITY

SECOND-ORDER COMPUTATIONAL HOMOGENISATION

MICRO-SCALE EQUILIBRIUM PROBLEM – LAGRANGE MULTIPLIERS

Principle of Multi-Scale Virtual Power

$$\underbrace{P : \delta F + Q : \delta G}_{\text{Macro}} = \frac{1}{V_\mu} \left[\int_{\Omega_\mu} P_\mu : (\delta F + \delta G \cdot Y + \nabla_Y \delta \tilde{u}) dV \right. \\
 - \delta L : \left(\int_{\partial \Omega_\mu} \tilde{u} \otimes N dA \right) - L : \left(\int_{\partial \Omega_\mu} \delta \tilde{u} \otimes N dA \right) \\
 - \delta M : \left(\int_{\Omega_\mu} (\nabla_Y \tilde{u} \otimes Y) \cdot J^{-1} dV \right) \\
 \left. - M : \left(\int_{\Omega_\mu} (\nabla_Y \delta \tilde{u} \otimes Y) \cdot J^{-1} dV \right) \right], \quad \forall (\delta F, \delta G, \delta \tilde{u}, \delta L, \delta M).$$

Micro-Scale Weak Equilibrium: $\delta F = 0, \delta G = 0$

$$\int_{\Omega_\mu} P_\mu : \nabla_Y \delta \tilde{u} dV - \delta L : \left(\int_{\partial \Omega_\mu} \tilde{u} \otimes N dA \right) \\
 - L : \left(\int_{\partial \Omega_\mu} \delta \tilde{u} \otimes N dA \right) \\
 - \delta M : \left(\int_{\Omega_\mu} (\nabla_Y \tilde{u} \otimes Y) \cdot J^{-1} dV \right) \\
 - M : \left(\int_{\Omega_\mu} (\nabla_Y \delta \tilde{u} \otimes Y) \cdot J^{-1} dV \right) = 0$$

Lagrange multiplier method is used to enforce the micro-scale kinematical constraints:

L enforces 1st constraint: $\int_{\partial \Omega_\mu} \tilde{u} \otimes N dA = 0$

M enforces 2nd constraint: $\int_{\Omega_\mu} (\nabla_Y \tilde{u} \otimes Y) \cdot J^{-1} dV = 0$

FEM discretisation
and linearisation

$$\begin{bmatrix} \mathbf{K}^{ii} & \mathbf{K}^{ib} & -\mathbf{C}_M^{i,T} & \mathbf{0} \\ \mathbf{K}^{bi} & \mathbf{K}^{bb} & -\mathbf{C}_M^{b,T} & -\mathbf{C}_L^T \\ \mathbf{C}_M^i & \mathbf{C}_M^b & \mathbf{0} & \mathbf{0} \\ \mathbf{0} & \mathbf{C}_L & \mathbf{0} & \mathbf{0} \end{bmatrix} \begin{Bmatrix} \Delta \tilde{u}^i \\ \Delta \tilde{u}^b \\ \Delta \lambda_M \\ \Delta \lambda_L \end{Bmatrix} = - \begin{Bmatrix} \mathbf{f}^i - \mathbf{C}_M^{i,T} \lambda_M \\ \mathbf{f}^b - \mathbf{C}_M^{b,T} \lambda_M - \mathbf{C}_L^T \lambda_L \\ \mathbf{C}_M \tilde{u} \\ \mathbf{C}_L \tilde{u}^b \end{Bmatrix}$$

SECOND-ORDER COMPUTATIONAL HOMOGENISATION

MACRO-SCALE CONSISTENT TANGENTS

The macroscopic consistent tangents are needed in a FE² framework

$$\frac{\partial \mathbf{r}}{\partial \mathbf{F}} = \mathbf{0} \Rightarrow \begin{bmatrix} \frac{\partial \mathbf{f}}{\partial \mathbf{F}} - \mathbf{C}^T \frac{\partial \tilde{\mathbf{u}}}{\partial \mathbf{F}} \\ \mathbf{C} \frac{\partial \tilde{\mathbf{u}}}{\partial \mathbf{F}} \end{bmatrix} = \mathbf{0}$$

$$\frac{\partial \mathbf{f}}{\partial \mathbf{F}} = \frac{\partial \mathbf{f}}{\partial \mathbf{u}} \frac{\partial \mathbf{u}}{\partial \mathbf{F}} = \mathbf{K} \left(\mathbf{D}^T + \frac{\partial \tilde{\mathbf{u}}}{\partial \mathbf{F}} \right)$$

$$\mathbf{u}_\mu = (\mathbf{F} - \mathbf{I}) \cdot \mathbf{Y} + \frac{1}{2} \mathbf{G} : (\mathbf{Y} \otimes \mathbf{Y}) + \tilde{\mathbf{u}}$$

↓

$$\mathbf{u}_\mu = \mathbf{D}^T (\mathbf{F} - \{\mathbf{I}\}) + \mathbf{V}^T \mathbf{G} + \tilde{\mathbf{u}}$$

(insertion discretisation)

$$\frac{\partial \mathbf{r}}{\partial \mathbf{G}} = \mathbf{0} \Rightarrow \begin{bmatrix} \frac{\partial \mathbf{f}}{\partial \mathbf{G}} - \mathbf{C}^T \frac{\partial \tilde{\mathbf{u}}}{\partial \mathbf{G}} \\ \mathbf{C} \frac{\partial \tilde{\mathbf{u}}}{\partial \mathbf{G}} \end{bmatrix} = \mathbf{0}$$

$$\frac{\partial \mathbf{f}}{\partial \mathbf{G}} = \frac{\partial \mathbf{f}}{\partial \mathbf{u}} \frac{\partial \mathbf{u}}{\partial \mathbf{G}} = \mathbf{K} \left(\mathbf{V}^T + \frac{\partial \tilde{\mathbf{u}}}{\partial \mathbf{G}} \right)$$

$$\begin{bmatrix} \mathbf{K} & -\mathbf{C}^T \\ \mathbf{C} & \mathbf{0} \end{bmatrix} \begin{bmatrix} \frac{\partial \tilde{\mathbf{u}}}{\partial \mathbf{F}} & \frac{\partial \tilde{\mathbf{u}}}{\partial \mathbf{G}} \\ \frac{\partial \lambda}{\partial \mathbf{F}} & \frac{\partial \lambda}{\partial \mathbf{G}} \end{bmatrix} = - \begin{bmatrix} \mathbf{K} \mathbf{D}^T & \mathbf{K} \mathbf{V}^T \\ \mathbf{0} & \mathbf{0} \end{bmatrix}$$

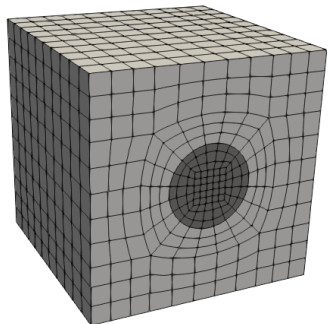
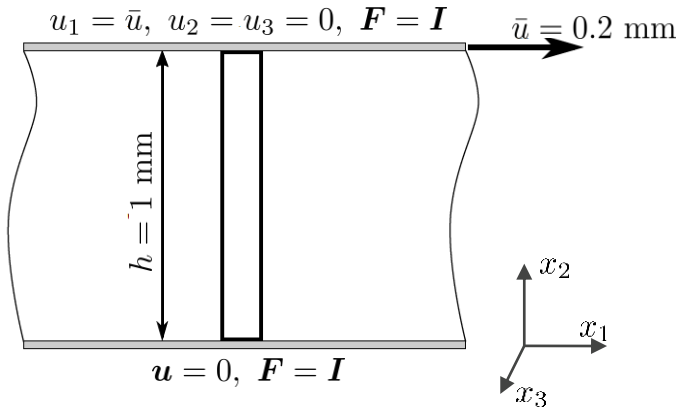
$$\mathbb{A} = \frac{\partial \mathbf{P}}{\partial \mathbf{F}} = \frac{\partial \lambda_L}{\partial \mathbf{F}} \quad \mathbb{A}_G = \frac{\partial \mathbf{P}}{\partial \mathbf{G}} = \frac{\partial \lambda_L}{\partial \mathbf{G}}$$

$$\mathbb{H}_F = \frac{\partial \mathbf{Q}}{\partial \mathbf{F}} = \frac{\partial \lambda_{Ms}}{\partial \mathbf{F}} \quad \mathbb{H} = \frac{\partial \mathbf{Q}}{\partial \mathbf{G}} = \frac{\partial \lambda_{Ms}}{\partial \mathbf{G}}$$

SECOND-ORDER COMPUTATIONAL HOMOGENISATION

NUMERICAL EXAMPLE – SIZE EFFECTS IN A BOUNDARY SHEAR LAYER

FE² 3D boundary shear layer simulation



Matrix
 $E_m = 70 \text{ GPa}$
 $\nu_m = 0.3$

Fibre
 $E_f = 700 \text{ GPa}$
 $\nu_f = 0.3$

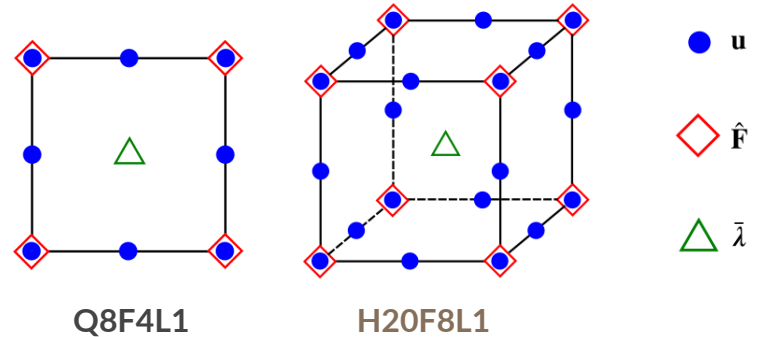
2410 Hexa20 elements

8 Gauss points

11088 nodes

The macro-scale second-gradient continuum must be discretized with elements that guarantee C^1 continuity:

- Mixed Finite Elements

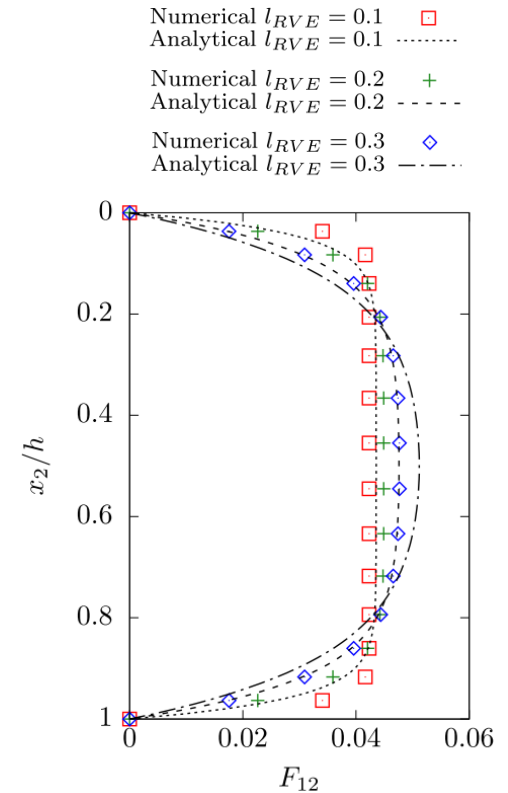
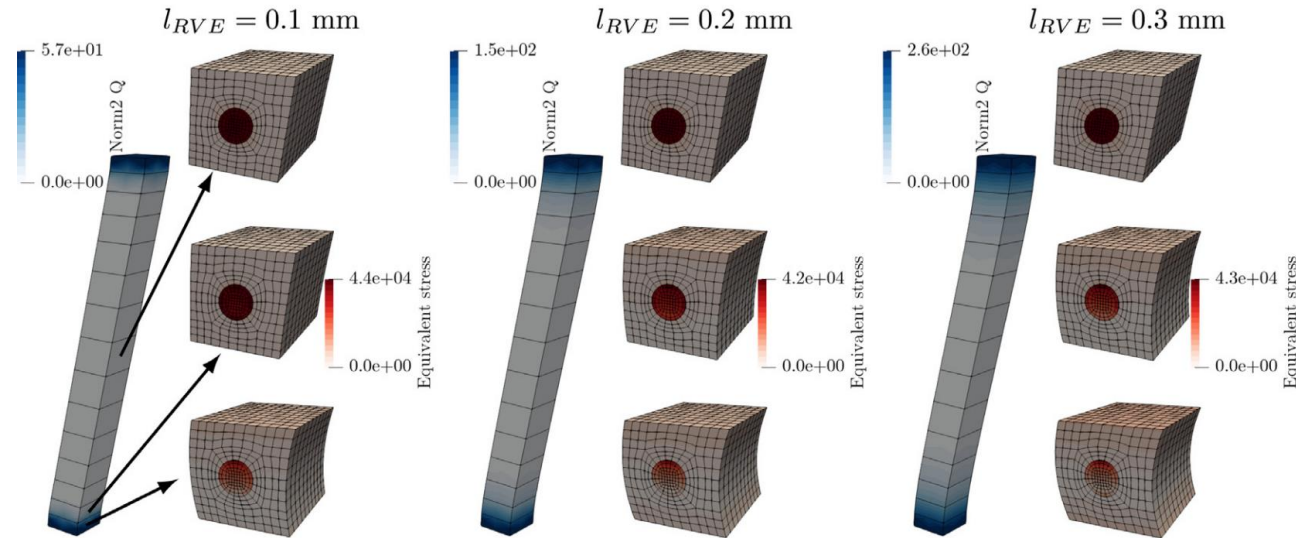


$$\int_{\Omega} \left(\mathbf{P} : \delta \mathbf{F} + \mathbf{Q} : \nabla_0^s \delta \hat{\mathbf{F}} - \delta \left[\boldsymbol{\lambda} : (\mathbf{F} - \hat{\mathbf{F}}) \right] \right) dV = \int_{\partial\Omega} \mathbf{t}_0 \cdot \delta \mathbf{u} dA + \int_{\partial\Omega} \mathbf{R}_0 : \delta \hat{\mathbf{F}} dA$$

$$\begin{bmatrix} \mathbf{K}^{uu} & \mathbf{K}^{uF} & \mathbf{K}^{u\lambda} \\ \mathbf{K}^{Fu} & \mathbf{K}^{FF} & \mathbf{K}^{F\lambda} \\ \mathbf{K}^{\lambda u} & \mathbf{K}^{\lambda F} & \mathbf{K}^{\lambda\lambda} \end{bmatrix}^{(k)} \begin{Bmatrix} \Delta \mathbf{u} \\ \Delta \hat{\mathbf{F}} \\ \Delta \boldsymbol{\lambda} \end{Bmatrix}^{(k+1)} = - \begin{Bmatrix} \mathbf{r}^u \\ \mathbf{r}^F \\ \mathbf{r}^\lambda \end{Bmatrix}^{(k)}$$

SECOND-ORDER COMPUTATIONAL HOMOGENISATION

NUMERICAL EXAMPLE – SIZE EFFECTS IN A BOUNDARY SHEAR LAYER



SECOND-ORDER COMPUTATIONAL HOMOGENISATION

NUMERICAL EXAMPLE – POLYCRYSTALLINE MATERIALS

M. Vieira de Carvalho, R. P. Cardoso Coelho, and F. M. A. Pires (2022), *Int. J. Numer. Methods Eng.*, 23, 21, 5155–5200
 I. A. Rodrigues Lopes, et al (2023), *Eur. J. Mech. - A/Solids*, 102, 105104.

Sandvik Nanoflex

Austenitic stainless steel

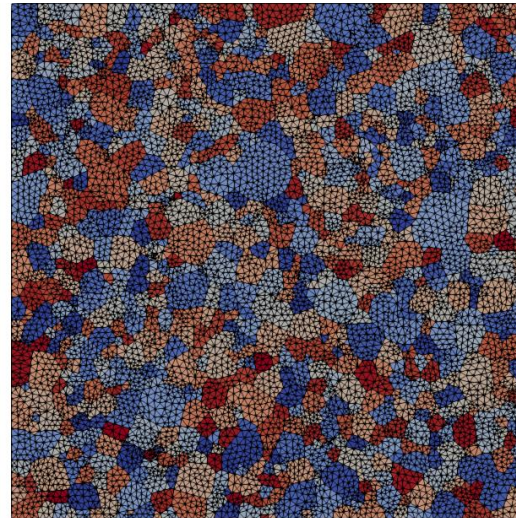
C_{11}	282.69	[GPa]
C_{12}	121.15	[GPa]
C_{44}	80.769	[GPa]
ϵ^{pa}	0.01	-
ϵ_0^{pa}	10	-
n_{visco}	8	-
m_{depth}	4	-
$\tilde{\tau}_y$	0.088	[GPa]
K	0.195	[GPa]
γ_0	0.01	-
m	0.6	-

Nadai-Ludwik power-law:

$$\tau_y(\gamma) = \tilde{\tau}_y + K (\gamma_0 + \gamma)^m$$

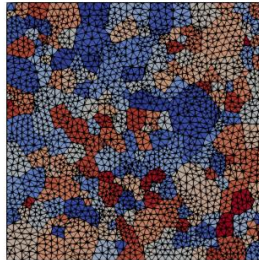
$$l_{RVE} = 0.8 \text{ mm}$$

1000 grains



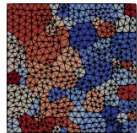
$$l_{RVE} = 0.4 \text{ mm}$$

250 grains



$$l_{RVE} = 0.2 \text{ mm}$$

65 grains



5 realizations for each RVE type

SECOND-ORDER COMPUTATIONAL HOMOGENISATION

NUMERICAL EXAMPLE – POLYCRYSTALLINE MATERIALS



F, G



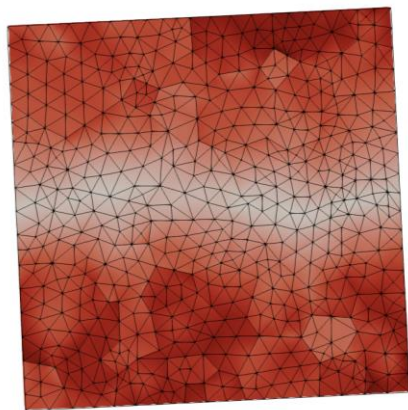
SECOND-ORDER COMPUTATIONAL HOMOGENISATION

NUMERICAL EXAMPLE – POLYCRYSTALLINE MATERIALS

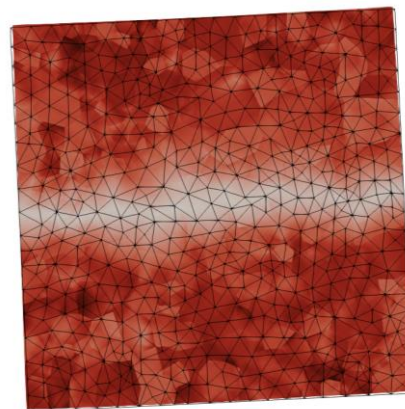
1st-order



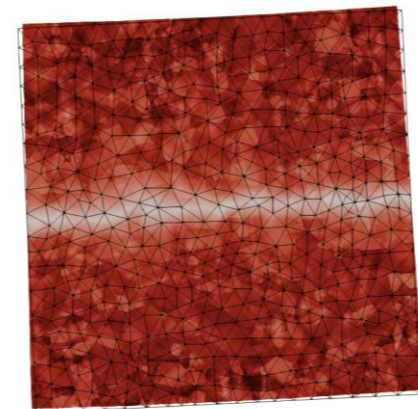
$l_{RVE} = 0.2$ mm



$l_{RVE} = 0.4$ mm



$l_{RVE} = 0.8$ mm

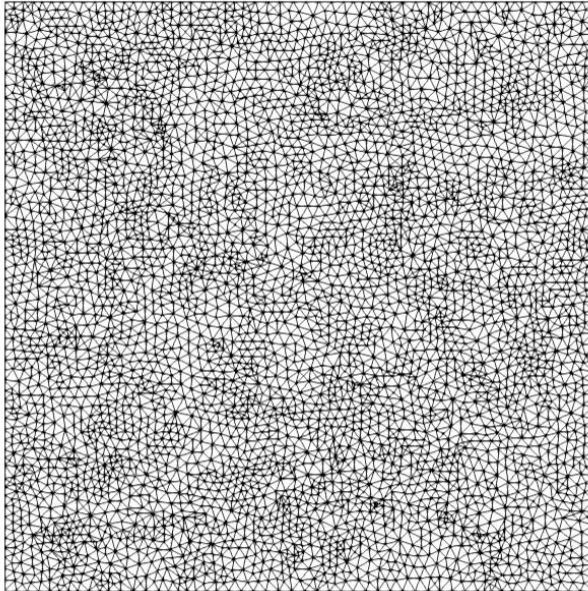


SECOND-ORDER COMPUTATIONAL HOMOGENISATION

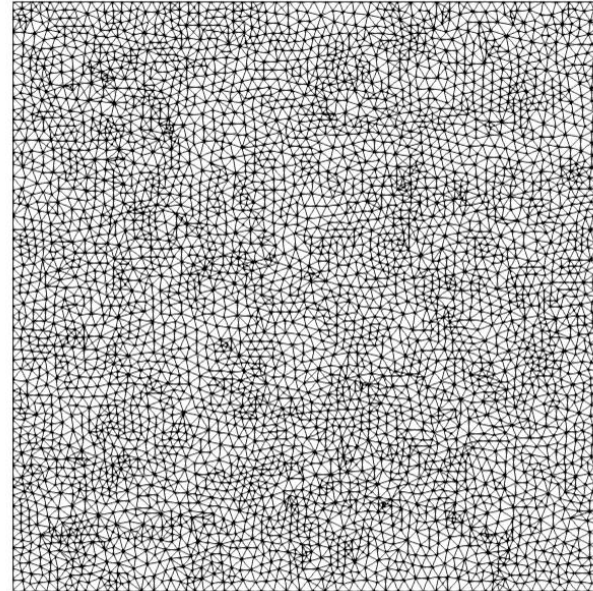
NUMERICAL EXAMPLE – MARTENSITIC TRANSFORMATION

$$l_{RVE} = 0.8 \text{ mm}$$

1st-order

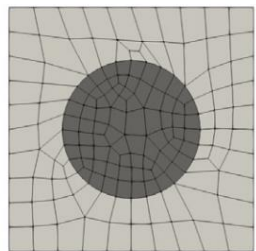
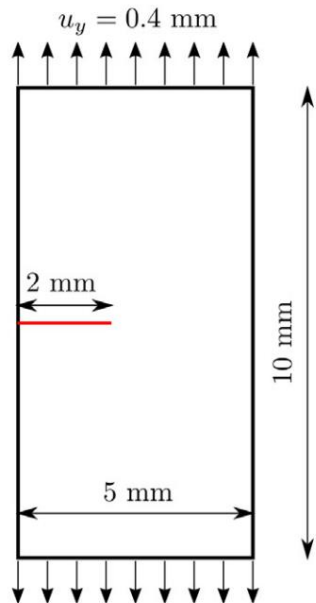


2nd-order

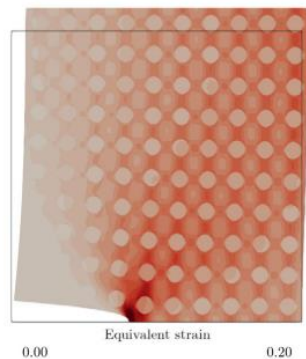


SECOND-ORDER COMPUTATIONAL HOMOGENISATION

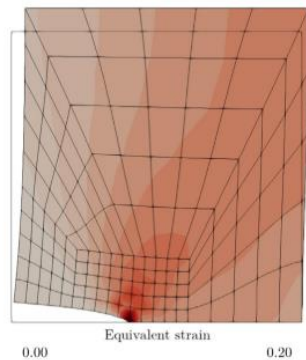
NUMERICAL EXAMPLE – STRAIN GRADIENTS AT CRACK TIP



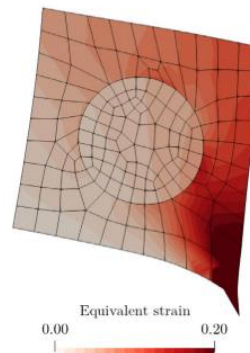
(b) Unit cell



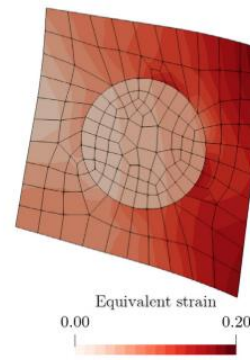
(a) DNS



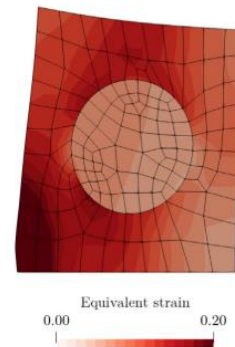
(c) Periodic



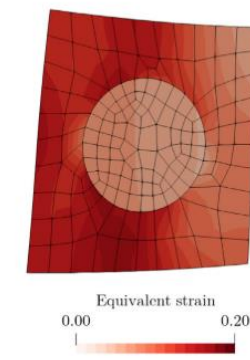
(a) DNS



(c) Periodic



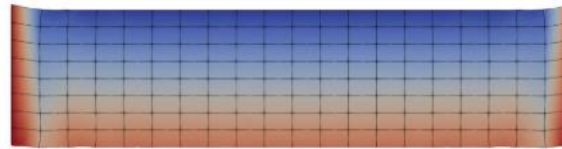
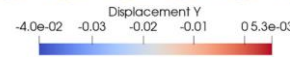
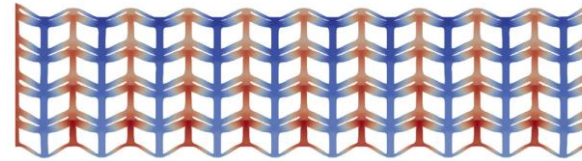
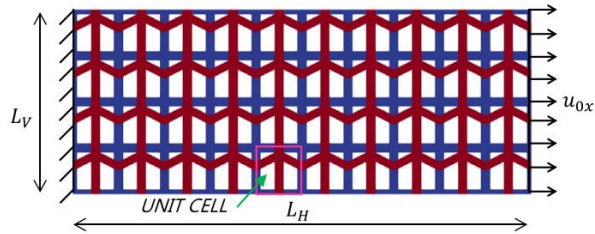
(a) DNS



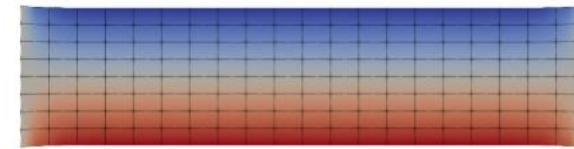
(c) Periodic

SECOND-ORDER COMPUTATIONAL HOMOGENISATION METAMATERIALS

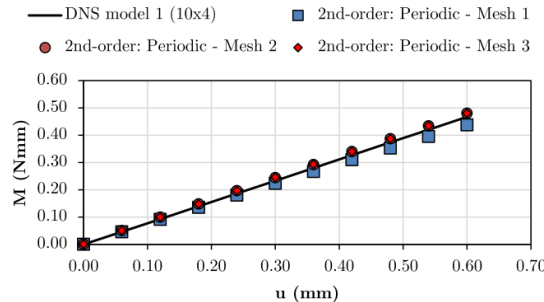
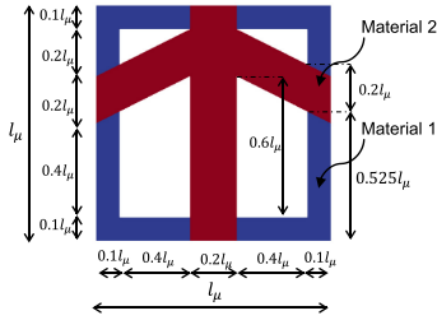
W. F. dos Santos, I. A. Rodrigues Lopes, F. M. Andrade Pires, and S. P. B. Proença (2023), *Comput. Methods Appl. Mech. Eng.*, 416, 16374.



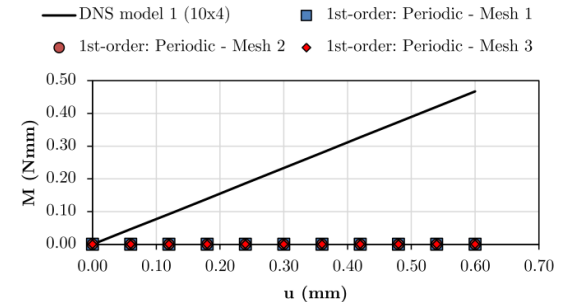
(c) Second-order: Periodic.



(d) First-order: Periodic.



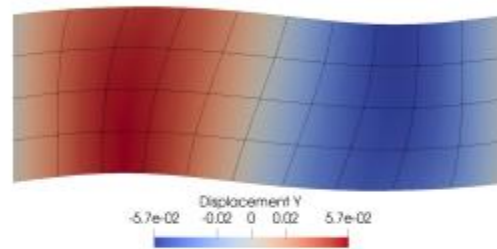
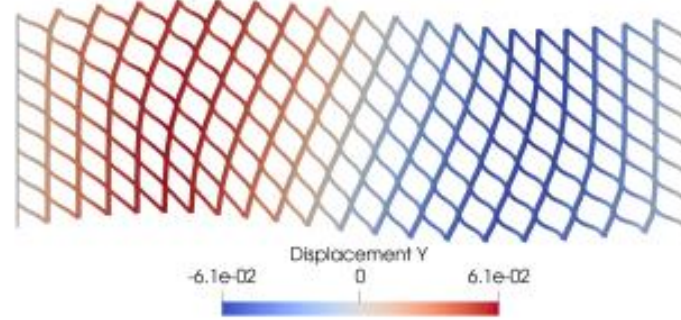
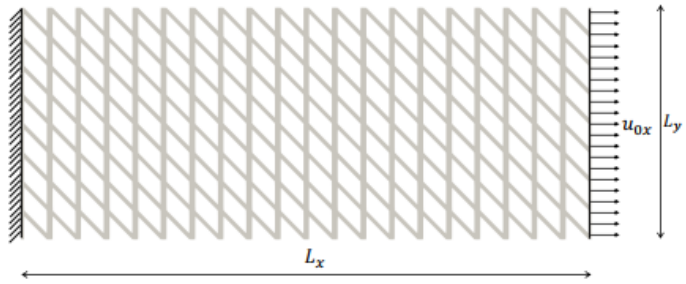
(c) Second-order: Periodic.



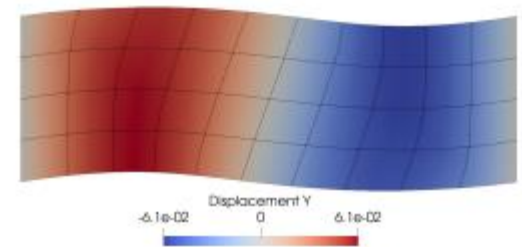
(d) First-order: Periodic.

SECOND-ORDER COMPUTATIONAL HOMOGENISATION METAMATERIALS

W. F. dos Santos, I. A. Rodrigues Lopes, F. M. Andrade Pires, and S. P. B. Proença (2024), Int. J. Solids Struc.



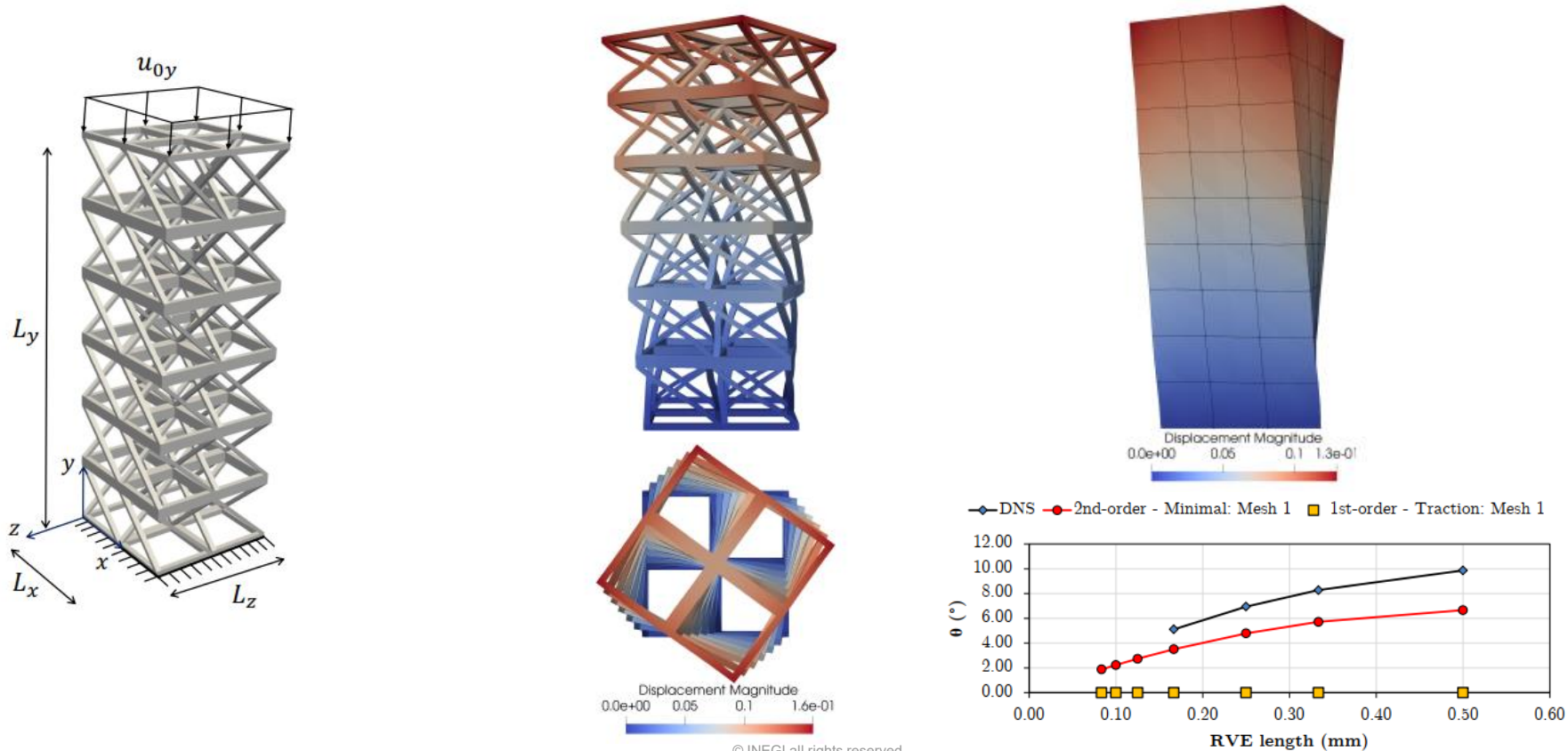
(b) Second-order: periodic.



(e) First-order: periodic.

SECOND-ORDER COMPUTATIONAL HOMOGENISATION METAMATERIALS

W. F. dos Santos, I. A. Rodrigues Lopes, F. M. Andrade Pires, and S. P. B. Proença (2024), Int. J. Solids Struc.

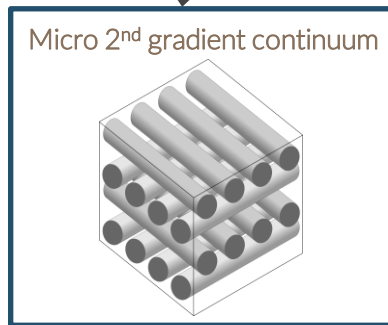


FULLY SECOND-ORDER COMPUTATIONAL HOMOGENISATION CONCEPT

I. A. Rodrigues Lopes and F. M. Andrade Pires (2022), *Int. J. Numer. Methods Eng.*, 123(21), 5274–5318



\mathbf{F}, \mathbf{G}



POSTULATED

Kinematic insertion

$$\mathbf{u}_\mu = (\mathbf{F} - \mathbf{I}) \mathbf{Y} + \frac{1}{2} \mathbf{G} : (\mathbf{Y} \otimes \mathbf{Y}) + \tilde{\mathbf{u}}$$

$$\hat{\mathbf{F}}_\mu = \mathbf{F} + \mathbf{G} \cdot \mathbf{Y} + \hat{\tilde{\mathbf{F}}}$$

Kinematic homogenisation

$$\mathbf{F} = \frac{1}{V_\mu} \int_{\Omega_\mu} \mathbf{F}_\mu d\Omega_\mu$$

$$\mathbf{G} = \frac{1}{V_\mu} \int_{\Omega_\mu} \mathbf{G}_\mu d\Omega_\mu$$

Kinematic admissibility

$$1^{\text{st}} \text{ Constraint: } \int_{\Omega_\mu} \nabla_Y \tilde{\mathbf{u}} dV = \mathbf{0} \Rightarrow \int_{\partial\Omega_\mu} \tilde{\mathbf{u}} \otimes \mathbf{N} dA = \mathbf{0}$$

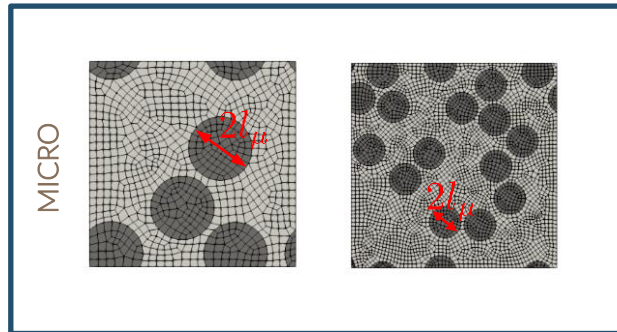
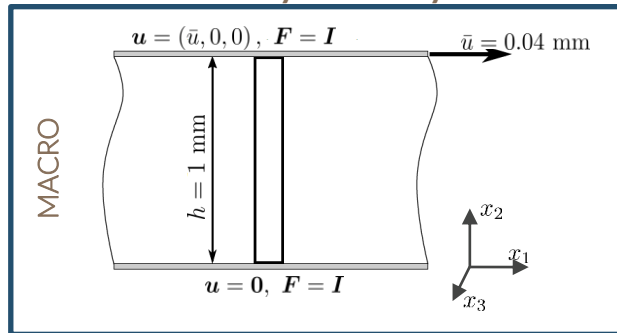
$$2^{\text{nd}} \text{ Constraint: } \int_{\Omega_\mu} \tilde{\mathbf{G}}_\mu dV = \mathbf{0} \Rightarrow \int_{\Omega_\mu} \nabla_Y^s \hat{\tilde{\mathbf{F}}} dV = \int_{\partial\Omega_\mu} \left(\hat{\tilde{\mathbf{F}}} \otimes \mathbf{N} \right)^S dA = \mathbf{0}$$

COMPATIBILITY

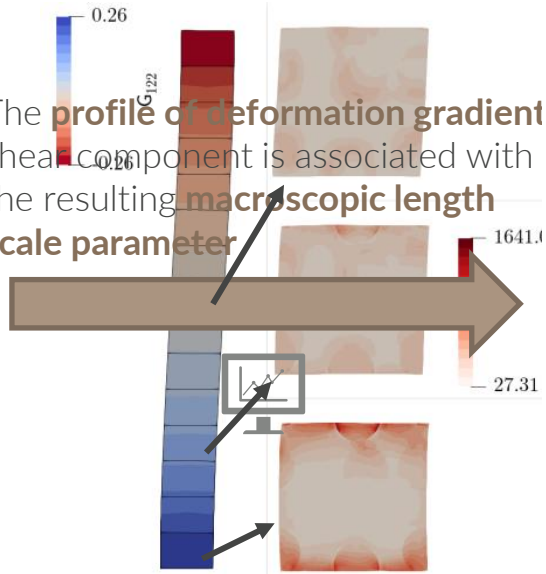
FULLY SECOND-ORDER COMPUTATIONAL HOMOGENISATION

ILLUSTRATION OF SIZE EFFECTS FROM THE MICRO-CONSTITUENTS

FE² boundary shear layer simulation



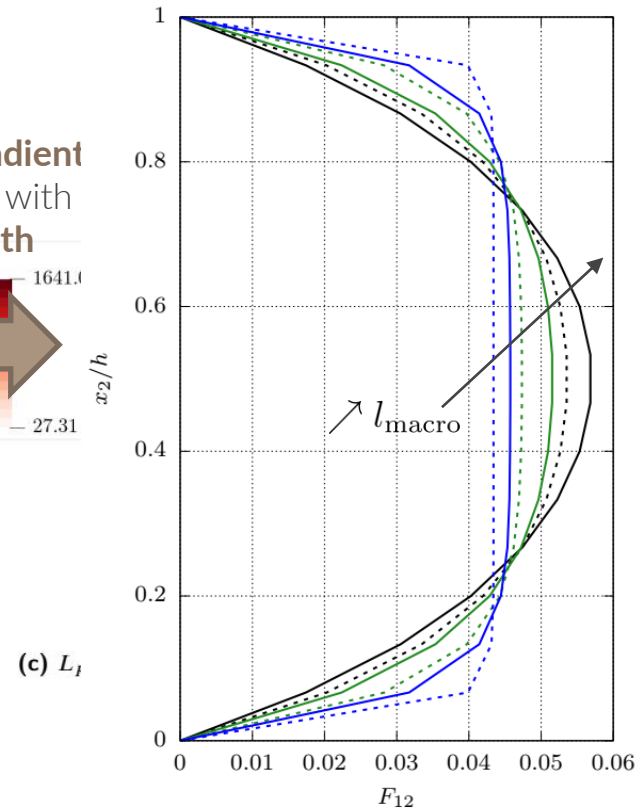
The profile of deformation gradient shear component is associated with the resulting macroscopic length scale parameter



$L_{RVE} = 0.828$ — 4 fibres —

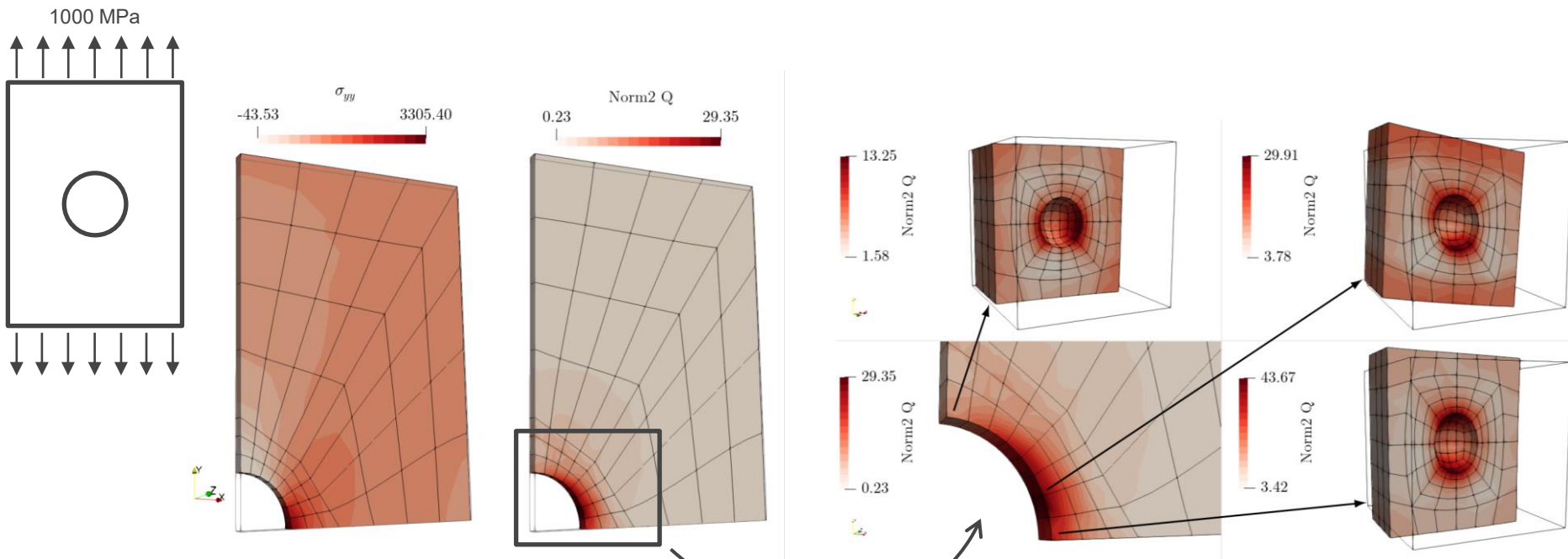
$L_{RVE} = 0.414$ — 16 fibres —

$L_{RVE} = 0.207$ — 16 fibres —



FULLY SECOND-ORDER COMPUTATIONAL HOMOGENISATION

3D MIXED ELEMENTS AT BOTH SCALES



Iteration	Relative residual (%)
1	5.86
2	0.53×10^{-2}
3	0.21×10^{-6}

I. A. Rodrigues Lopes and F. M. Andrade Pires (2022),
Int. J. Numer. Methods Eng., 123(21), 5274–5318

COMPUTATIONAL HOMOGENISATION WITH FRACTURE



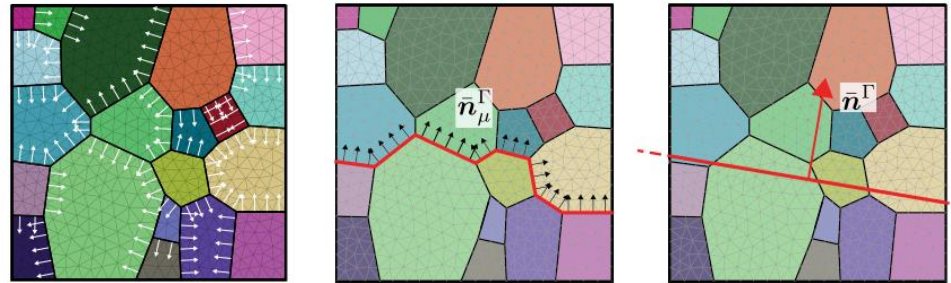
COMPUTATIONAL HOMOGENISATION WITH COHESIVE ELEMENTS

APPLICATION TO INTERGRANULAR FRACTURE

First-order computational homogenisation is enhanced with the presence of strong discontinuities

$$\mathbf{F} = \frac{1}{|\Omega_\mu|} \int_{\Omega_\mu} \mathbf{F}_\mu \, dV + \frac{1}{|\Omega_\mu|} \int_{\Gamma_\mu} \llbracket \tilde{\mathbf{u}}_\mu \rrbracket \otimes \bar{\mathbf{n}}_\mu^\Gamma \, dA.$$

An equivalent macroscopic crack is determined from the micro-cracks orientation



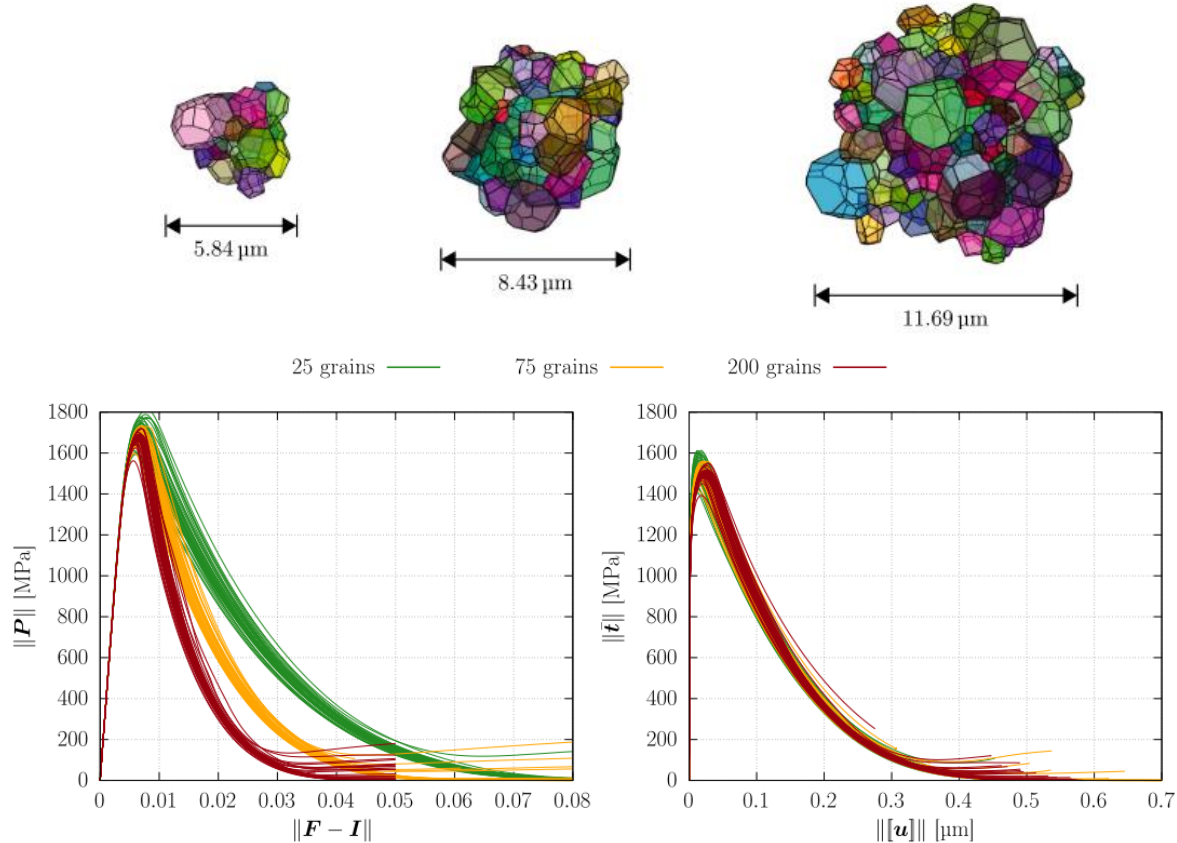
Macroscopic cohesive traction determined as the projection of the macro-stress on the macro-crack

$$\bar{\mathbf{t}} \equiv \mathbf{P} \bar{\mathbf{n}}^\Gamma$$

COMPUTATIONAL HOMOGENISATION WITH COHESIVE ELEMENTS

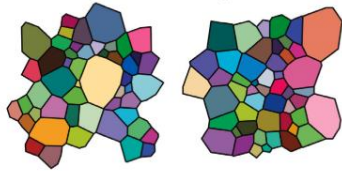
APPLICATION TO INTERGRANULAR FRACTURE

M. Vieira de Carvalho, I. A. Rodrigues Lopes, and F. M. Andrade Pires (2023), *Int. J. Plast.*, 71, 103780.

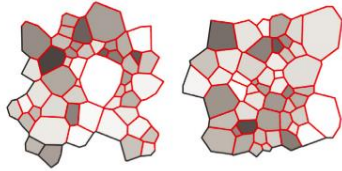


COMPUTATIONAL HOMOGENISATION WITH COHESIVE ELEMENTS

APPLICATION TO INTERGRANULAR FRACTURE



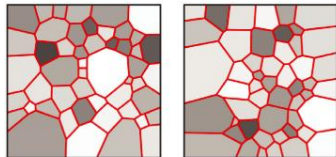
(a) Periodic tessellations.



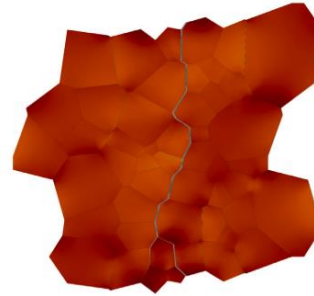
(c) Periodic cohesive elements.



(b) Non-periodic tessellations.



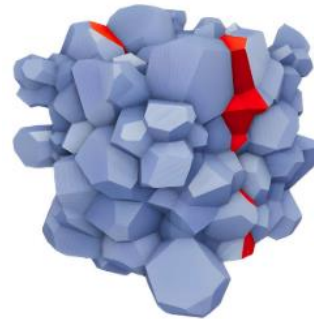
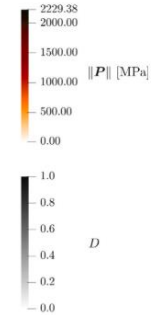
(d) Non-periodic cohesive elements.



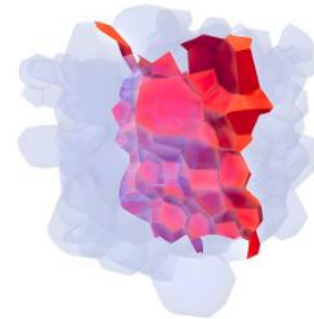
(a) $F_{xx} = 1.01$.



(b) $F_{xx} = 1.05$.



(a) Fracture surface.

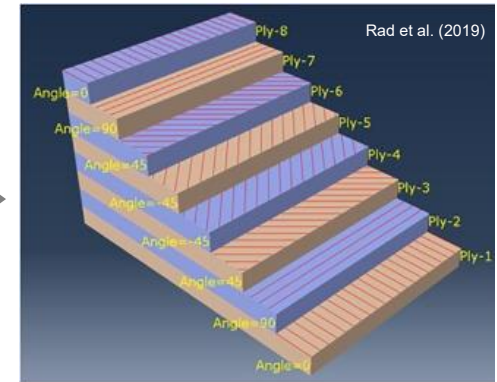
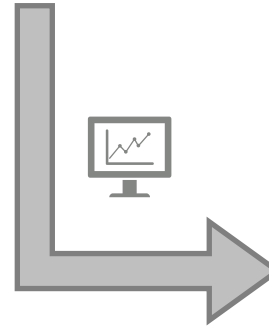
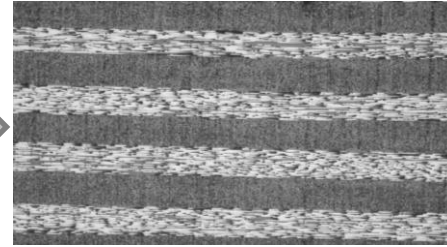
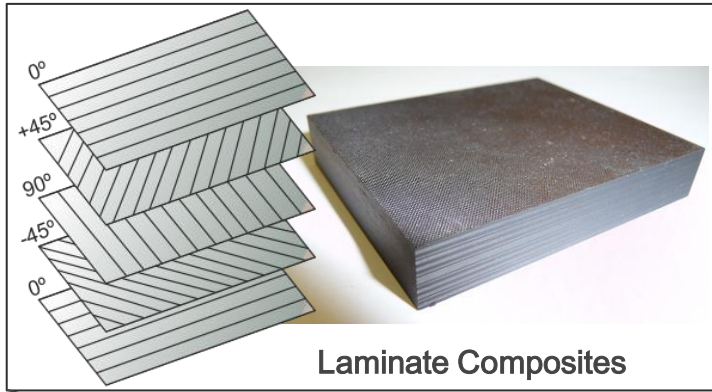


(b) Fracture surface highlighted.

MESO-SCALE MODELLING OF COMPOSITES



MODELLING COMPOSITES AT THE MESO-SCALE CONCEPT



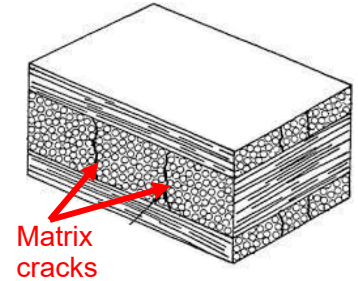
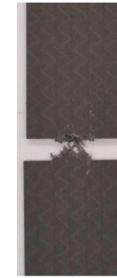
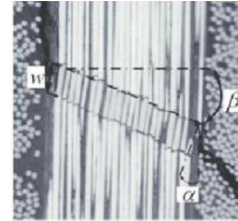
Meso-scale models
(at the ply level)

MODELLING COMPOSITES AT THE MESO-SCALE REQUIREMENTS

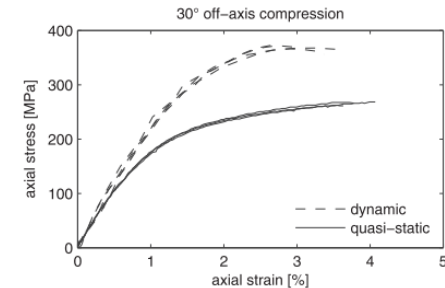
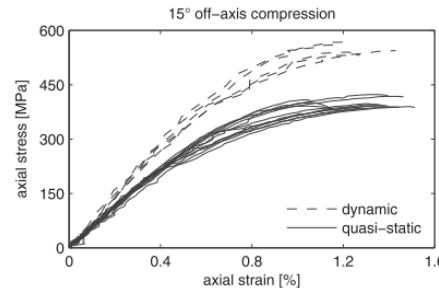


What effects to be included in the models?

- **Transverse isotropy** (elastic and plastic domains)
- **Damage Mechanisms**
 - Matrix
 - Fibres
- **Visco-elasticity**
- **Visco-plasticity**
- **Finite strains**



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3D INVARIANT-BASED VISCO-ELASTIC – VISCO-PLASTIC MODEL

TRANSVERSE ISOTROPIC STRESS-STRAIN RELATIONSHIP AND VISCO-ELASTICITY

Elastic stress-strain relationship

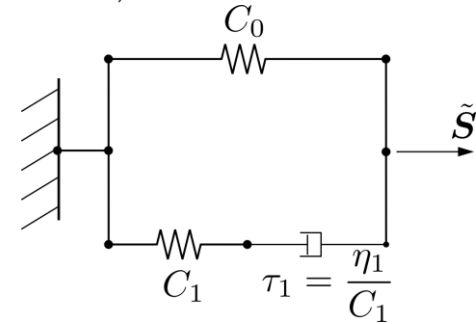
$$\tilde{\mathbf{S}}_0 = \lambda \operatorname{tr}(\tilde{\mathbf{E}}^e) \mathbf{I} + 2\mu_T \tilde{\mathbf{E}}^e + \alpha \left[\operatorname{tr}(\tilde{\mathbf{A}}\tilde{\mathbf{E}}^e) + \operatorname{tr}(\tilde{\mathbf{E}}^e) \tilde{\mathbf{A}} \right] \mathbf{I} + 2(\mu_L - \mu_T) (\tilde{\mathbf{E}}^e \tilde{\mathbf{A}} + \tilde{\mathbf{A}} \tilde{\mathbf{E}}^e) + \beta \operatorname{tr}(\tilde{\mathbf{A}}\tilde{\mathbf{E}}^e) \tilde{\mathbf{A}}$$

Structural Tensor: $\tilde{\mathbf{A}} = \tilde{\mathbf{a}} \otimes \tilde{\mathbf{a}}$

Elastic Green-Lagrange: $\tilde{\mathbf{E}}^e = \frac{1}{2} (\tilde{\mathbf{C}}^e - \mathbf{I})$

Elastic Parameters: $\begin{Bmatrix} \lambda \\ \alpha \\ \beta \\ \mu_T \\ \mu_L \end{Bmatrix} \leftrightarrow \begin{Bmatrix} E_{11} \\ E_{22} \\ G_{12} \\ \nu_{12} \\ \nu_{23} \end{Bmatrix}$

Rheological model: 1 Hooke element and 1 Maxwell element (Gerbaud et al., 2019)



Finite strain theory of visco-elasticity (Kalske, 2000)

$$\tilde{\mathbf{S}} = \tilde{\mathbf{S}}_0 + \tilde{\mathbf{S}}_1$$

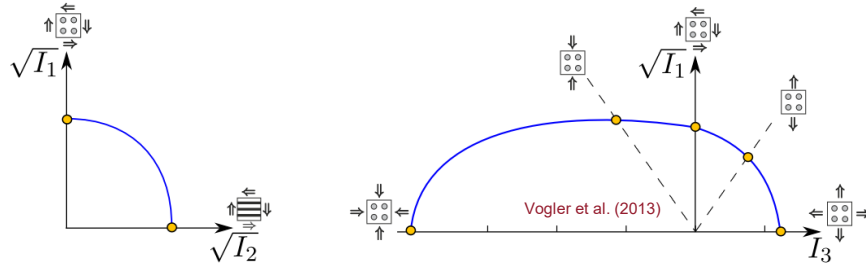
$$\tilde{\mathbf{S}}_1^{n+1} = e^{-\frac{\Delta t}{\tau_1}} \tilde{\mathbf{S}}_1^n + \frac{1 - e^{-\frac{\Delta t}{\tau_1}}}{\frac{\Delta t}{\tau_1}} (\tilde{\mathbf{S}}_{0,n+1}^{\text{pind}} - \tilde{\mathbf{S}}_{0,n}^{\text{pind}})$$

3D INVARIANT-BASED VISCO-ELASTIC – VISCO-PLASTIC MODEL

TRANSVERSELY ISOTROPIC YIELD FUNCTION AND PLASTIC POTENTIAL

Yield function based on invariants

$$f\left(\tilde{\Sigma}_s, \tilde{\mathbf{A}}, \tilde{\varepsilon}^p\right) = \alpha_1 I_1 + \alpha_2 I_2 + \alpha_3 I_3 + \alpha_{32} I_3^2 - 1 \leq 0$$



$$I_1 = \frac{1}{2} \left(\text{tr} \tilde{\Sigma}_s^{\text{pind}} \right)^2 - \text{tr} \left(\tilde{\mathbf{A}} \left[\tilde{\Sigma}_s^{\text{pind}} \right]^2 \right),$$

$$I_2 = \text{tr} \left(\tilde{\mathbf{A}} \left[\tilde{\Sigma}_s^{\text{pind}} \right]^2 \right),$$

$$I_3 = \text{tr} \tilde{\Sigma}_s - \text{tr} \left(\tilde{\mathbf{A}} \tilde{\Sigma}_s \right)$$

Hardening parameters, α_i , determined from experimental curves obtained for different stress states

$$\alpha_1(\tilde{\varepsilon}^p), \alpha_2(\tilde{\varepsilon}^p), \alpha_3(\tilde{\varepsilon}^p), \alpha_{32}(\tilde{\varepsilon}^p), \dot{\tilde{\varepsilon}}^p = \sqrt{\frac{1}{2} \tilde{\mathbf{D}}^p : \tilde{\mathbf{D}}^p}$$

Non-associative visco-plastic potential

$$g\left(\tilde{\Sigma}_s, \tilde{\mathbf{A}}\right) = \beta_1 I_1 + \beta_2 I_2 + \beta_{32} I_3^2 - 1 = \frac{1}{2} \tilde{\Sigma}_s : \mathbb{M} : \tilde{\Sigma}_s$$

Parameters (β_i) calibrated to obtained a given plastic Poisson's ratio (Vogler et al., 2013)

Flow tensor and rate of plastic deformation:

$$\mathbf{N}_g = \frac{\partial g\left(\tilde{\Sigma}_s, \tilde{\mathbf{A}}\right)}{\partial \tilde{\Sigma}_s} = \mathbb{M} : \tilde{\Sigma}_s.$$

$$\tilde{\mathbf{D}}^p = \dot{\gamma} \mathbf{N}_g$$

Perzyna model employed to describe the evolution of the plastic multiplier rate

$$\dot{\gamma} = \frac{\left\langle f^m\left(\tilde{\Sigma}_s, \tilde{\mathbf{A}}, \tilde{\varepsilon}^p\right) \right\rangle}{\eta}$$

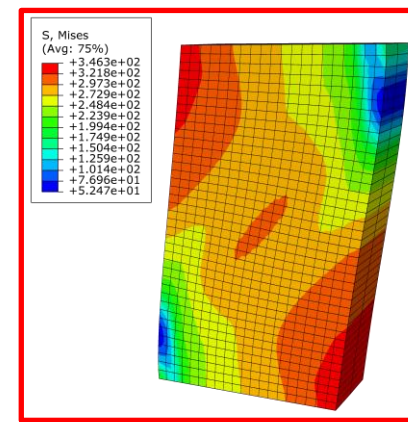
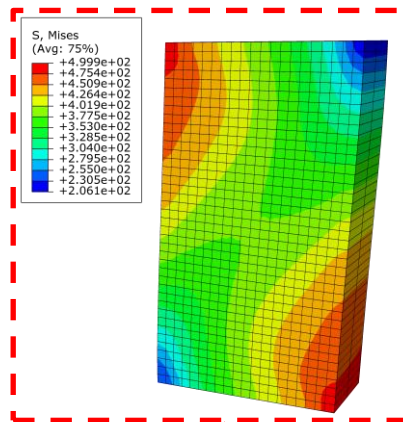
3D INVARIANT-BASED VISCO-ELASTIC – VISCO-PLASTIC MODEL

NNUMERICAL EXAMPLES – OFF-AXIS COMPRESSION

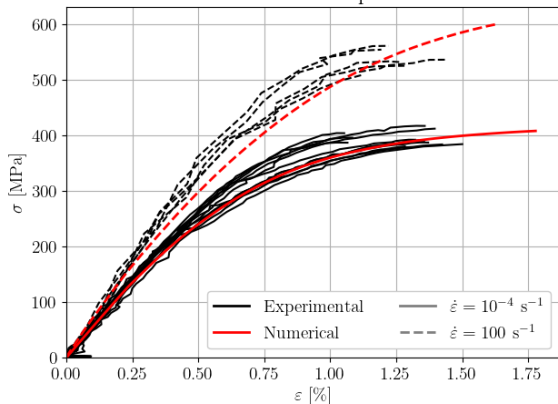
HexPly IM7-8552 (carbon fibre in epoxy matrix)

Constitutive parameters calibrated by
[Vogler et al., 2013; Koerber, et al., 2018, Gerbaud et al. 2019]

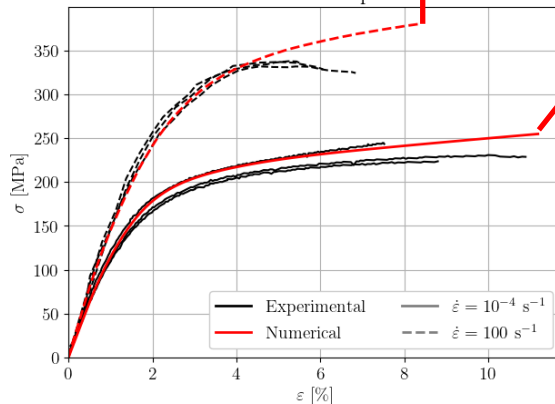
Quasi-static tests (10^{-4} s^{-1}) and dynamic tests (100 s^{-1})
[Koerber, et al., 2010].



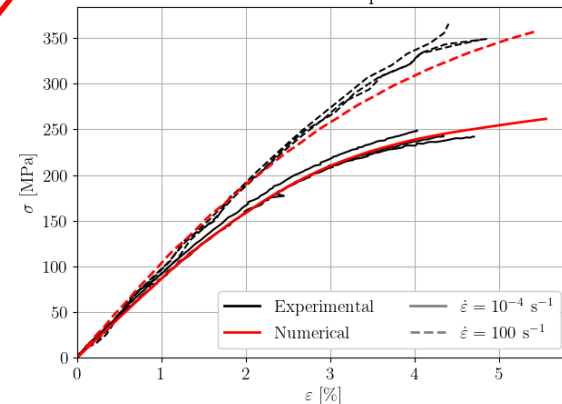
IM7 8552 off-axis compression 15°



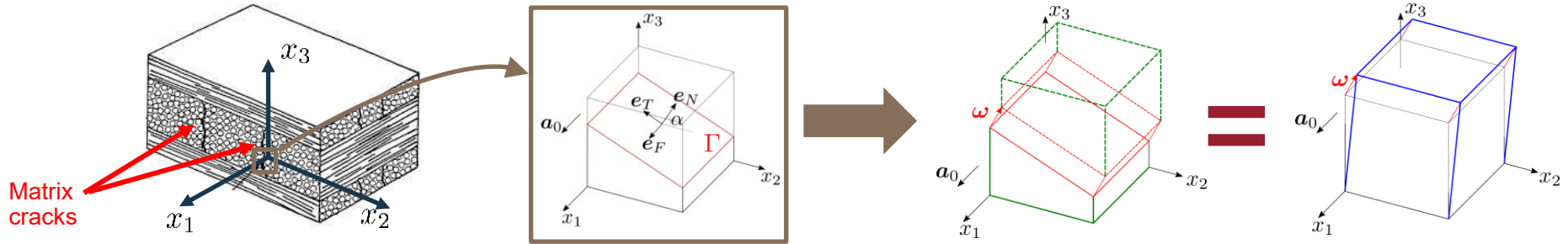
IM7 8552 off-axis compression 45°



IM7 8552 off-axis compression 90°



FINITE STRAIN SMEARED-CRACK MODEL CONCEPT



Transverse failure criterion [Camanho et al., 2015]

$$f_T(\boldsymbol{\sigma}, \mathbf{A}) = \zeta_1 \bar{I}_1 + \zeta_2 \bar{I}_2 + \zeta_3 \bar{I}_3 + \zeta_{32} \bar{I}_3^2 - 1 \leq 0$$

Additive Deformation Gradient Decomposition [Leone, 2015]

$$\mathbf{F}_B + \nabla \mathbf{u}_c = \mathbf{F}$$

Based on homogenisation considerations...

$$\mathbf{F}_B = \mathbf{F} - \frac{|\Gamma|}{V_B} \mathbf{R}_\alpha \boldsymbol{\omega} \otimes \mathbf{e}_N$$

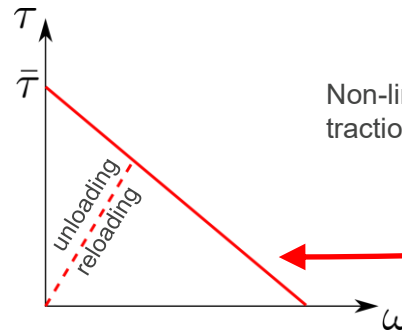
Non-linear equilibrium between bulk stress and cohesive tractions:

$$\mathbf{P}(\mathbf{F}_B) \cdot \mathbf{e}_N = \mathbf{R}_\alpha \cdot \boldsymbol{\tau}(\boldsymbol{\omega})$$

Cohesive law – defines cohesive tractions on the crack

$$\begin{cases} \text{normal} \\ \tau_N = (1-d) |\bar{\tau}_N|, & \text{if } \omega_N \geq \omega_N^{max} \\ \tau_N = (1-d) |\bar{\tau}_N| \frac{\omega_N}{\omega_N^{max}}, & \text{if } 0 \leq \omega_N < \omega_N^{max} \\ \tau_N = k_N \omega_N, & \text{if } \omega_N \leq 0 \end{cases}$$

$$\begin{cases} \text{shear} \\ \tau_s = (1-d) \bar{\tau}_s, & \text{if } \omega_s \geq \omega_s^{max} \\ \tau_s = (1-d) \bar{\tau}_s \frac{\omega_s}{\omega_s^{max}}, & \text{if } \omega_s < \omega_s^{max} \end{cases}$$



LONGITUDINAL FAILURE

A CONTINUUM DAMAGE APPROACH

Longitudinal failure criterion

$$f_L = \phi_L - r_L \leq 0$$

$$\phi_L = \begin{cases} \frac{E_1}{X_T} E_{B,11}, & \text{if } E_{B,11} \geq 0 \\ \frac{E_1}{X_C} E_{B,11}, & \text{if } E_{B,11} < 0 \end{cases}$$

$$r_L = \max(1, \max(\phi_L))$$

Damaged stress-strain relationship [Maimi, 2007]

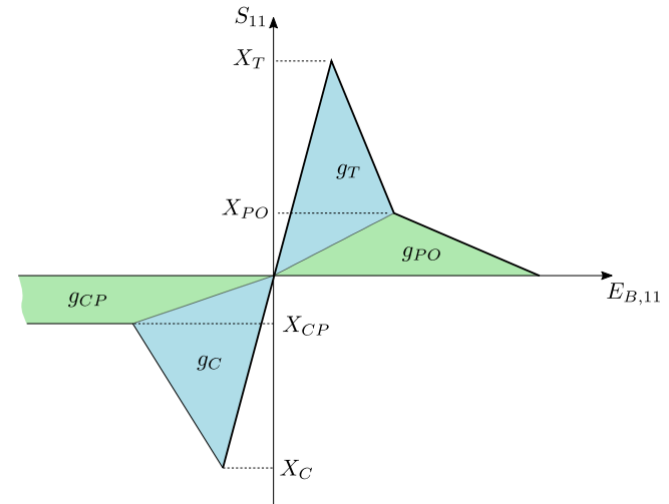
$$\mathbf{E}_B = \mathbf{H}\mathbf{S} \Leftrightarrow \mathbf{S} = \mathbf{H}^{-1}\mathbf{E}_B$$

$$\mathbf{H} = \begin{bmatrix} \frac{1}{E_{11}(1-d_1)} & -\frac{\nu_{21}}{E_{22}} & -\frac{\nu_{21}}{E_{22}} & 0 & 0 & 0 \\ \frac{1}{E_{22}} & \frac{1}{E_{22}} & -\frac{\nu_{23}}{E_{22}} & 0 & 0 & 0 \\ \frac{1}{E_{22}} & -\frac{\nu_{23}}{E_{22}} & \frac{1}{E_{22}} & 0 & 0 & 0 \\ 0 & 0 & 0 & \frac{1}{G_{12}} & 0 & 0 \\ 0 & 0 & 0 & 0 & \frac{1}{G_{23}} & 0 \\ 0 & 0 & 0 & 0 & 0 & \frac{1}{G_{12}} \end{bmatrix}$$

sym

Evolution of the damage variable

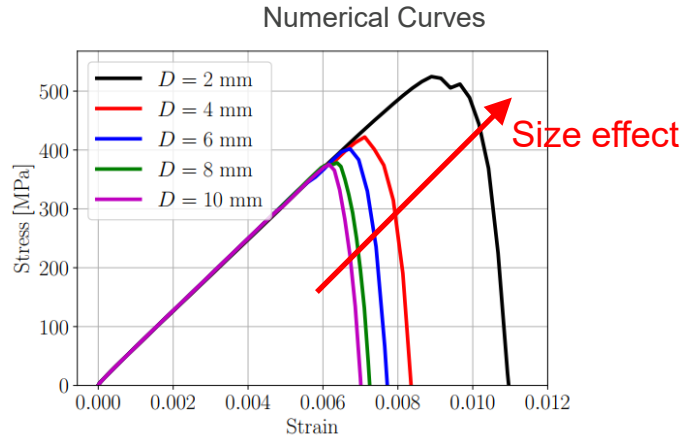
Crack band model [Bažant, 1983]



FINITE STRAIN SMEARED-CRACK MODEL

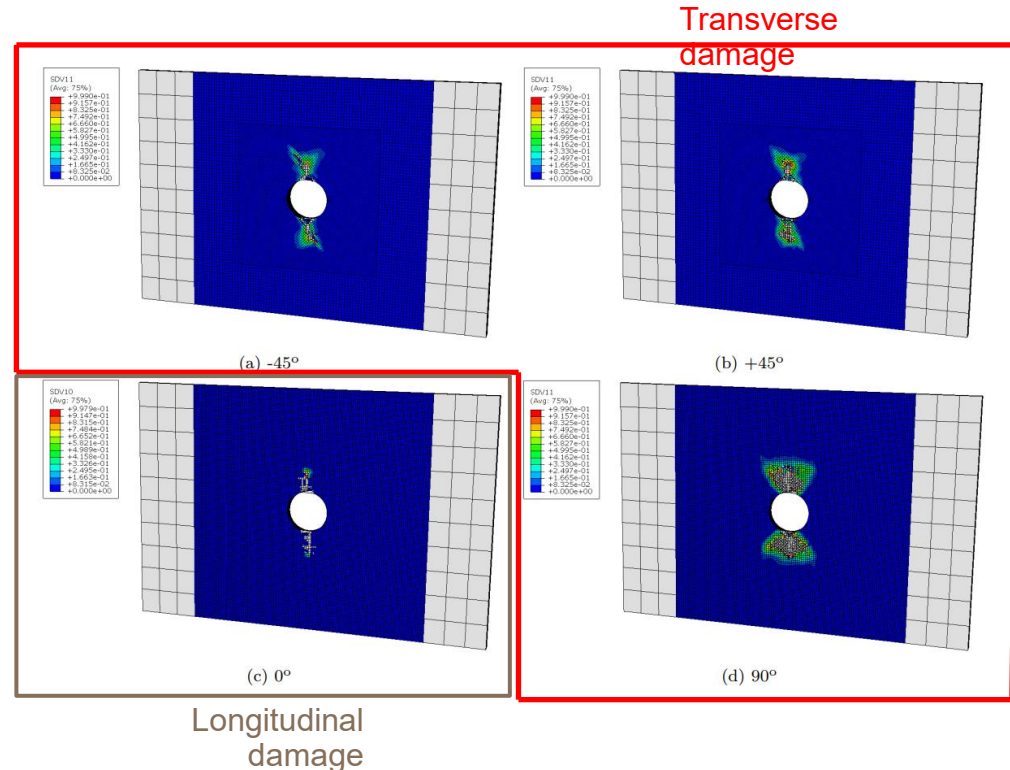
NUMERICAL RESULTS – QI-OPEN-HOLE

Quasi-isotropic [90/0/+45]_{3s} open-hole tension
HexPly IM7-8552



Strength comparison with experimental data (Camanho et al., 2007)

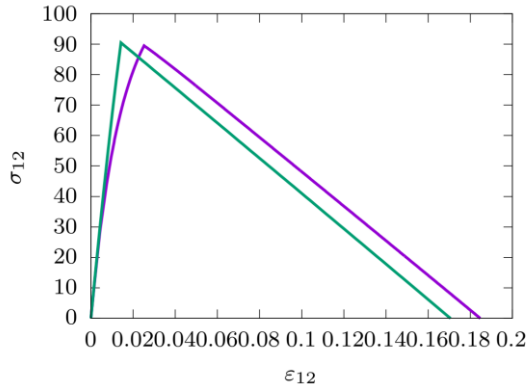
Hole diameter (mm)	Strength (MPa)		
	Experimental	Numerical	
2	555.7	524.7	-5.6%
4	480.6	422.1	-12.2%
6	438.7	402.1	-8.3%
8	375.7	378.2	0.7%
10	373.7	375.3	0.4%



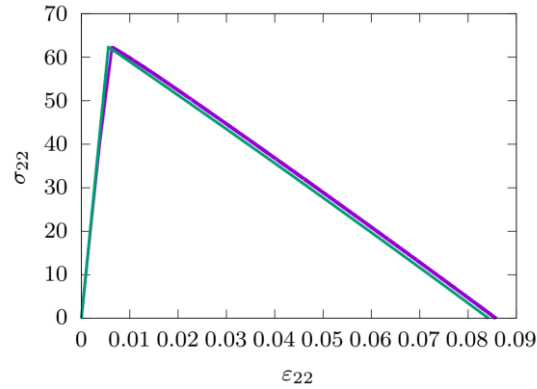
FINITE STRAIN SMEARED-CRACK MODEL

COUPLING WITH VISCO-ELASTIC – VISCO-PLASTIC MODEL

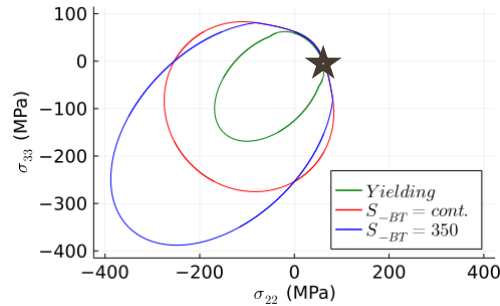
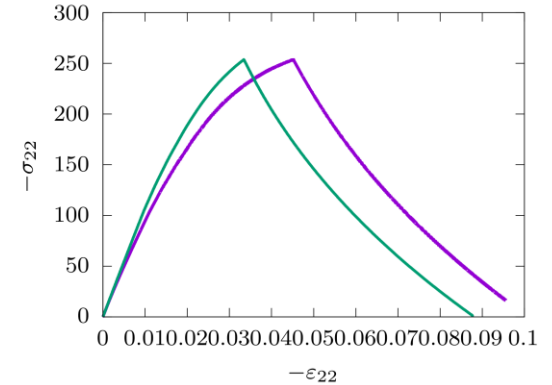
Longitudinal Shear



Transverse Tension



Transverse Compression



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- BSC/Severo Ochoa Mobility Program – Incoming Grant
- Clean Sky 2 Joint Undertaking (JU) – **TREAL** project — Thermoplastic material allowable generation using a reliability-based virtual modeling platform (Grant agreement No. 864723)
- European Union’s Horizon 2020 research and innovation programme – **DIDEAROT** project – Digital strategies for robust manufacturing and design of composite aircraft (Grant agreement No. 101056682)
- Fundação para a Ciência e a Tecnologia – PhD Grant: SFRH/BD/100093/2014

Multi-scale Modelling of Complex Materials undergoing Strain Gradients and Damage

Severo Ochoa Seminar

Igor A. Rodrigues Lopes
ilopes@inegi.up.pt

INSTITUTE OF SCIENCE AND INNOVATION IN
MECHANICAL AND INDUSTRIAL ENGINEERING

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